



Low Vacuum Scanning Electron Microscopy and Microanalysis

***Principles and Practice of Variable Pressure/Environmental
Scanning Electron Microscopy (VP-ESEM), Debbie J Stokes,
John Wiley&Sons 2008***
Several graphs courtesy of prof. Bradley Thiel



- Conventional scanning electron microscopy (SEM) is a widely used as an analytical tool. However, there are several limitations on the type of samples, which may be observed.
 - When a specimen consists of a non-conductive material, or not properly grounded to the specimen stub, charging occurs. **Charging is the build-up of an excess of electrons on the surface of the specimen, which causes many undesirable artefacts.** Both secondary and backscattered electrons utilised for SEM observations are significantly affected.
- **The elimination of specimen charging is accomplished in the LowVac/ESEM scanning electron microscope by the introduction of a gas, water, nitrogen or air into the specimen chamber.**



Outline

- **Introduction to Variable Pressure (Environmental) Scanning Electron Microscopy**
- **Image formation in VP-SEM**
- **X-ray Microanalysis in VP-SEM**
- **Variable pressure method**
- **EBSD in VP-SEM**
- **Conclusions**

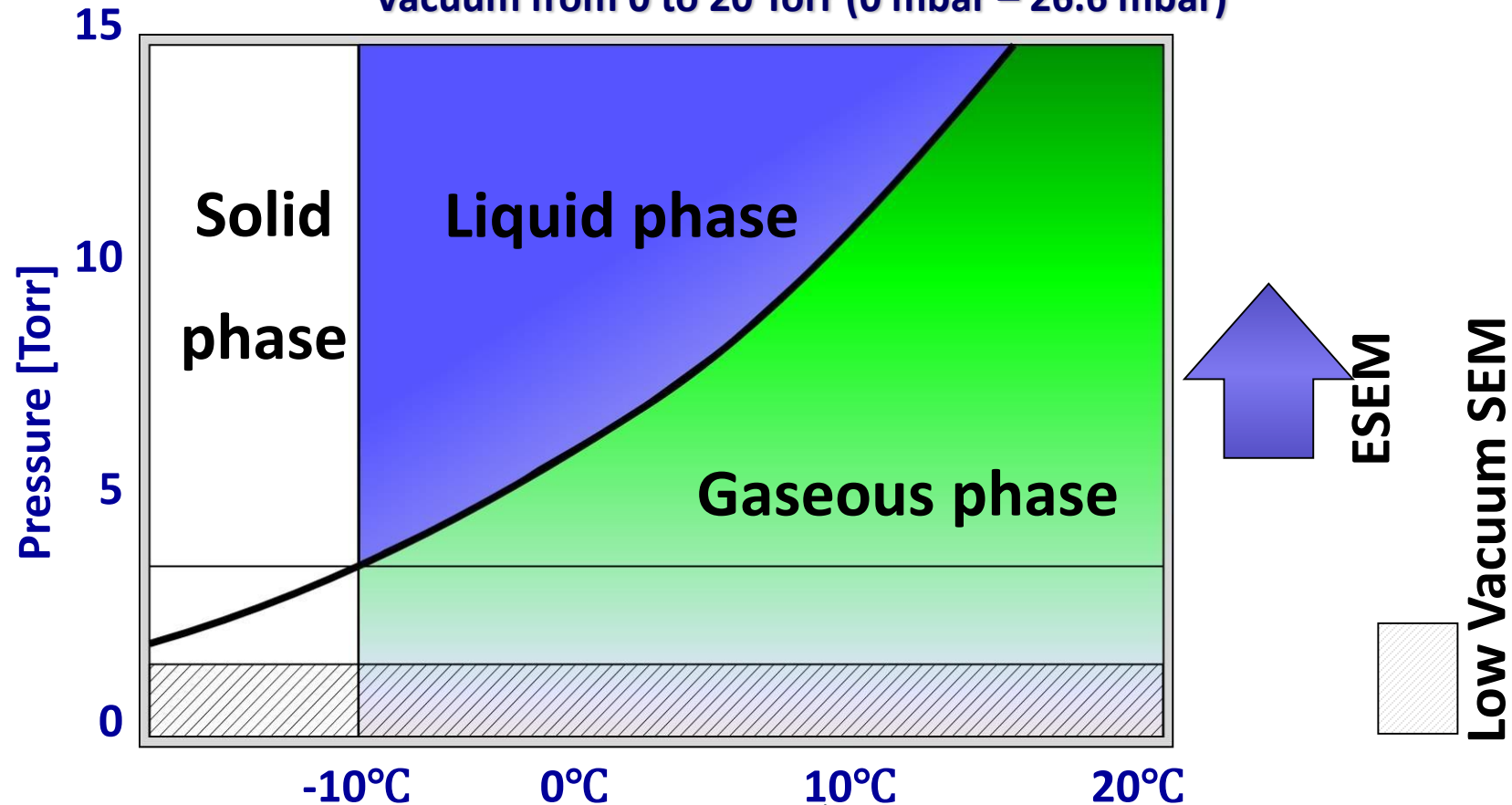


➤ LV-SEM – Low Vacuum Scanning Electron Microscopy

Vacuum from 0 to 1 Torr (0 mbar – 1.33 mbar)

➤ VP-SEM or Environmental Scanning Electron Microscopy

Vacuum from 0 to 20 Torr (0 mbar – 26.6 mbar)

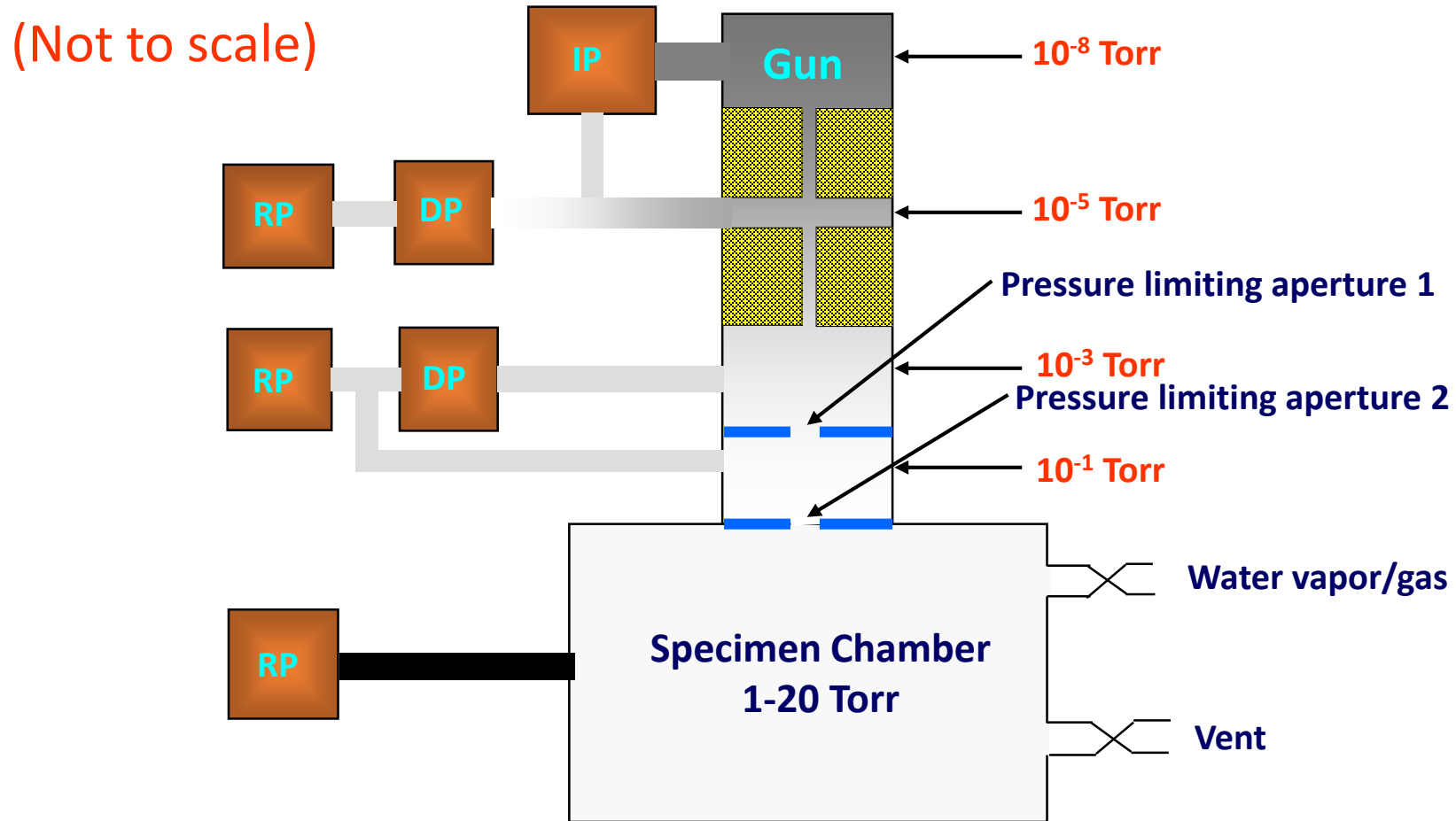


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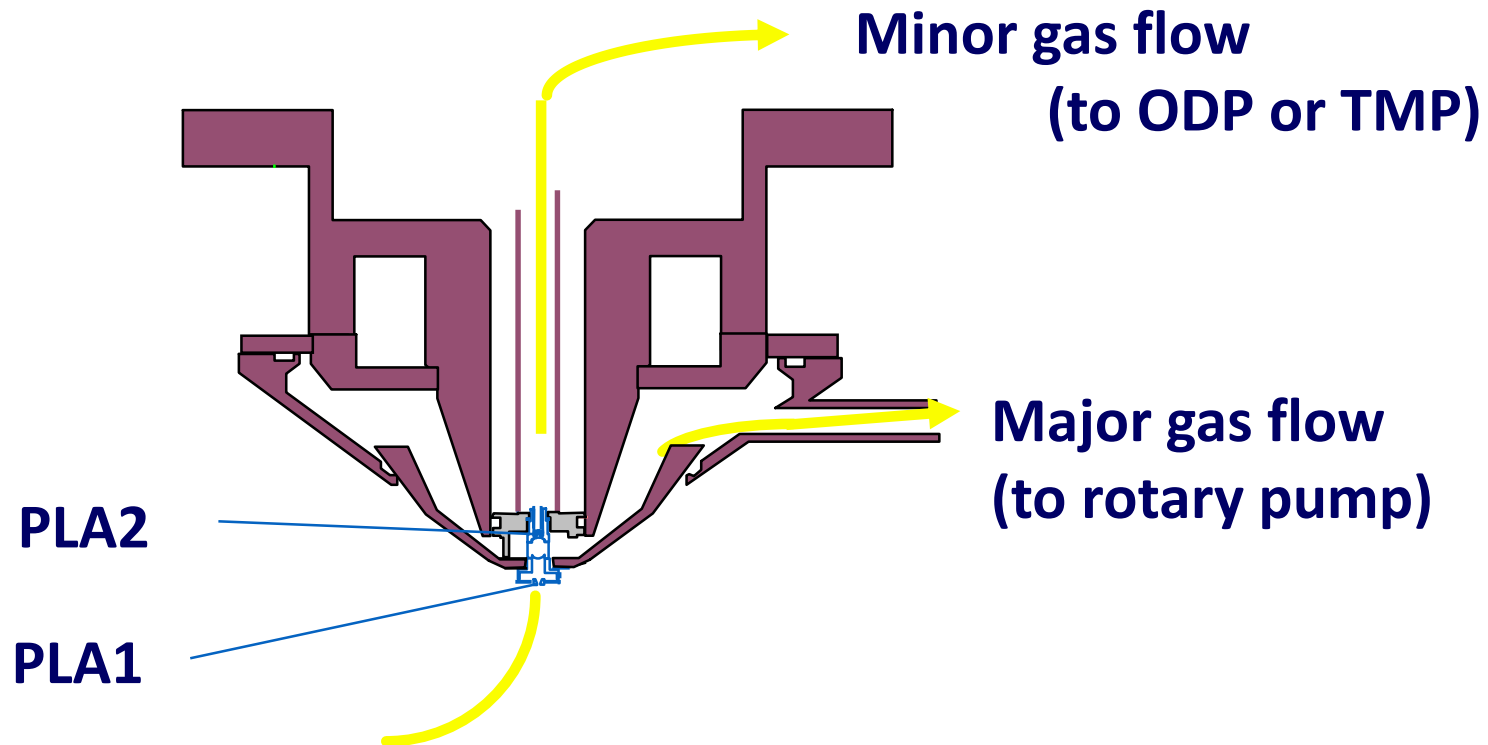
Variable Pressure Scanning Electron Microscope





Pumping system of ESEM

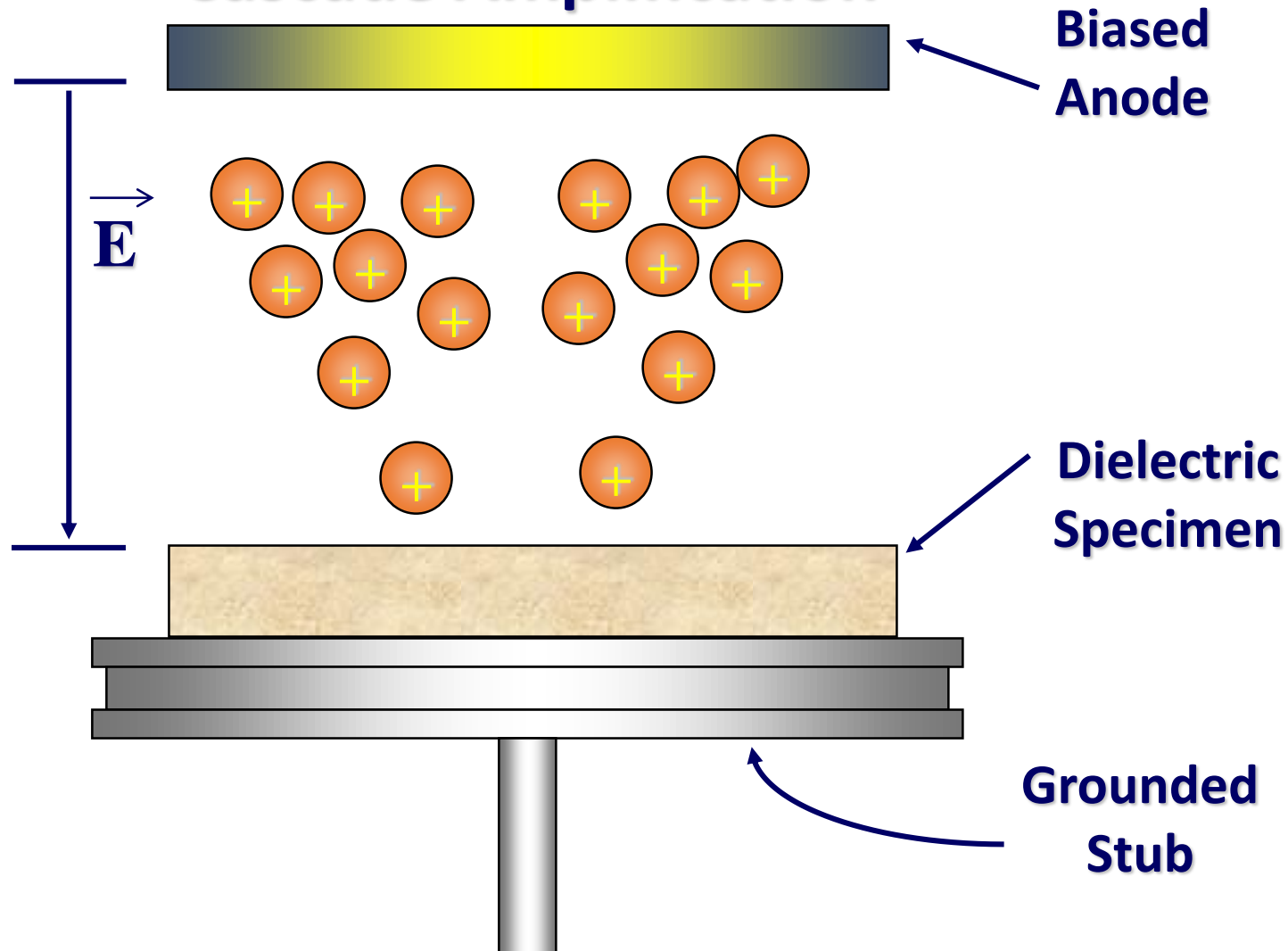
Emission area: min. 10^{-5} Torr



Gas flow from chamber
max. 20 Torr



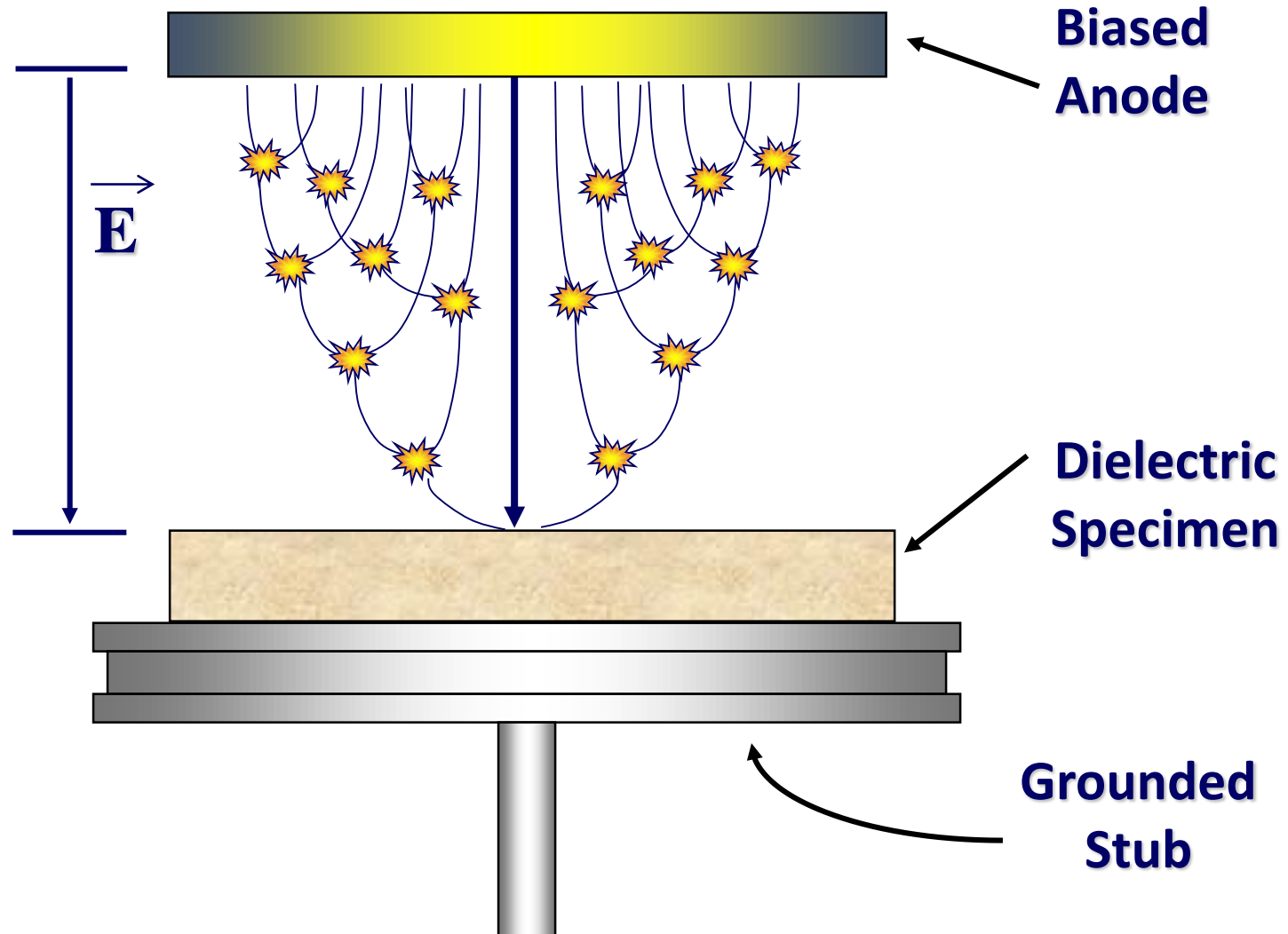
Cascade Amplification





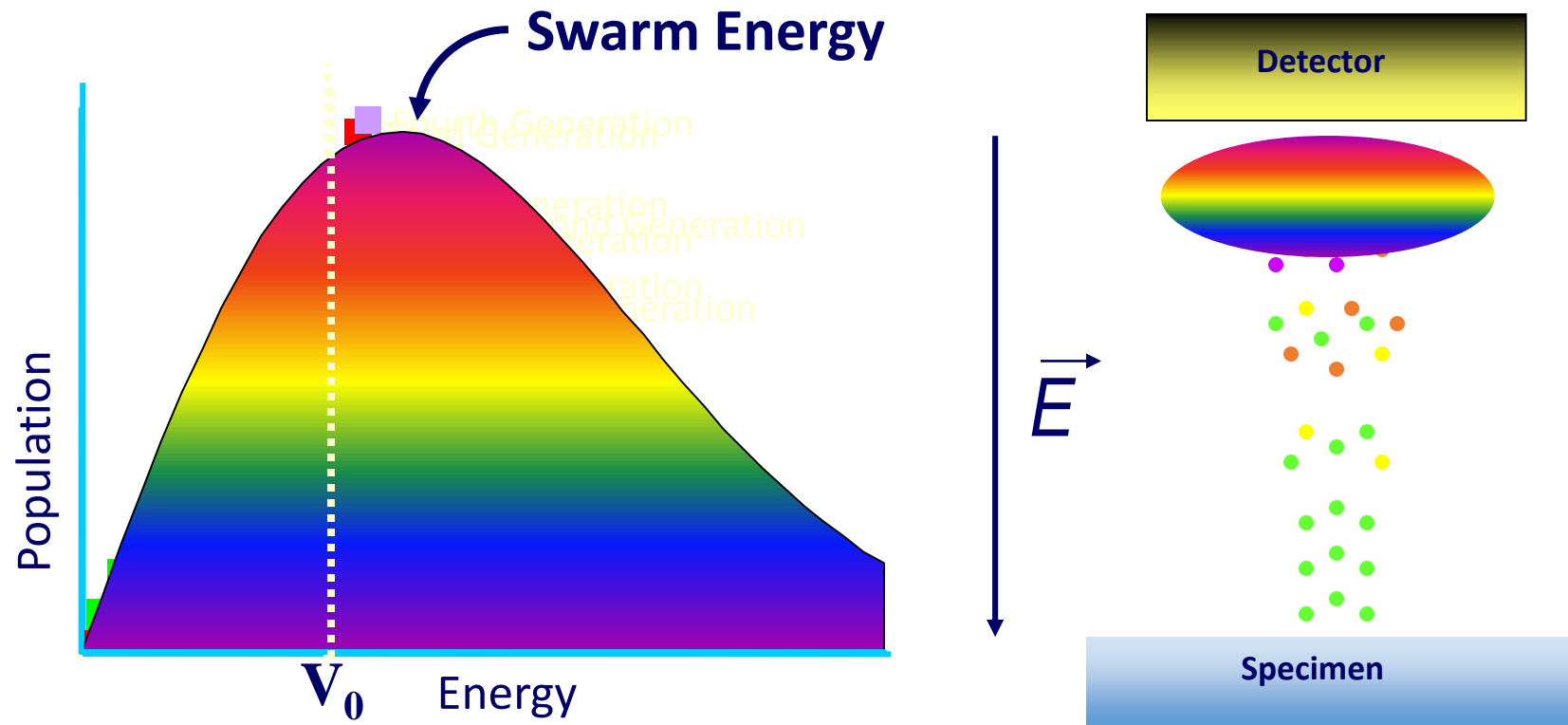
Cascade Amplification

The SEs collide with the gas molecules in the chamber and create further electrons which are also accelerated in the electric field: a cascade or avalanche promotes amplification



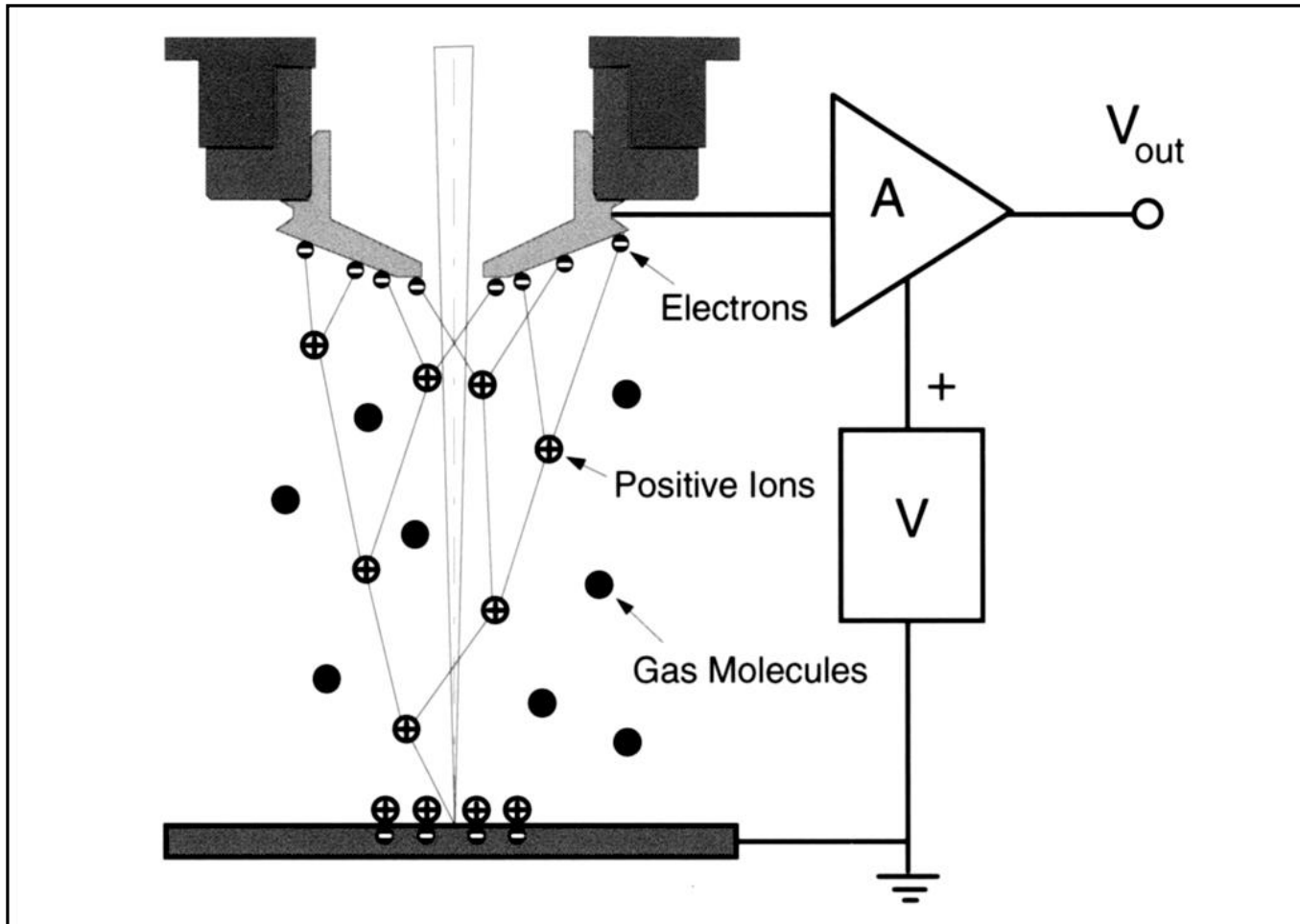


SE Amplification





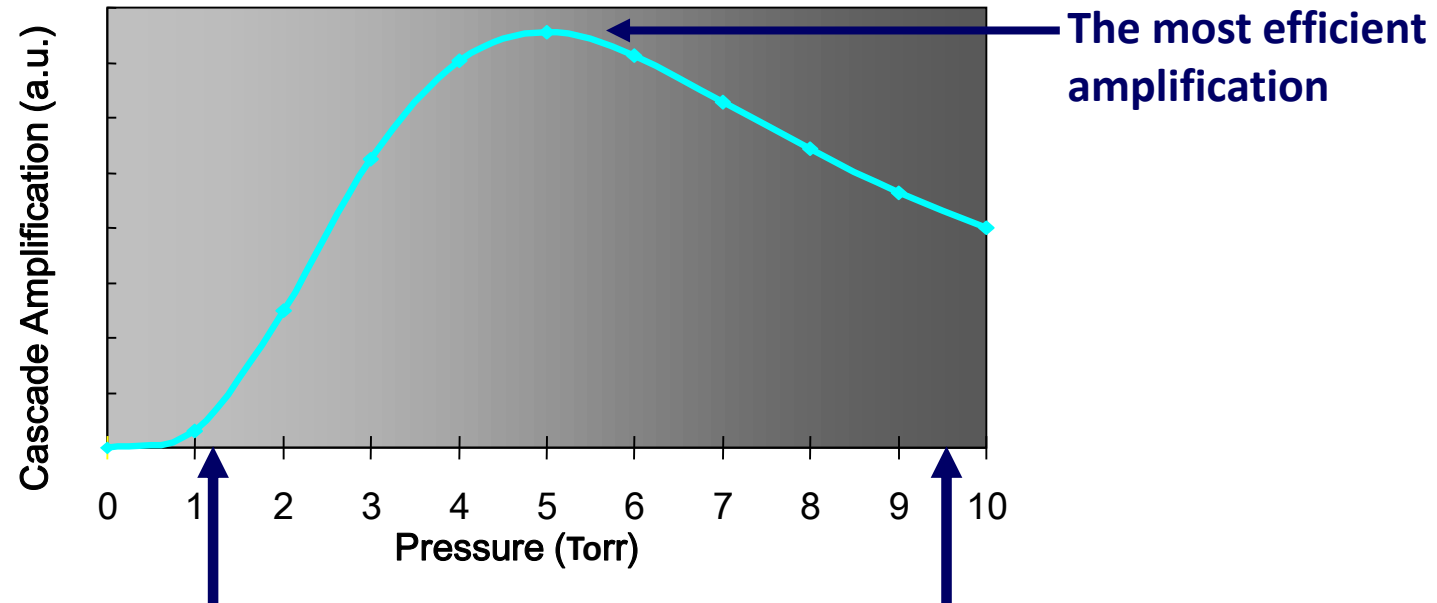
A simplified schematic diagram of gaseous signal amplification via ionisation



- The primary electron beam impinges on the sample, leading to the production of backscattered and secondary electrons.
- The electric field between the positively biased anode and the grounded specimen stage accelerates secondary electrons towards the anode.
- Secondary electrons collide with and ionise gas molecules in their path and amplify the secondary signal.
- Positive ions drift towards the sample surface to compensate negative charge build up.



General Shape of Amplification Curve



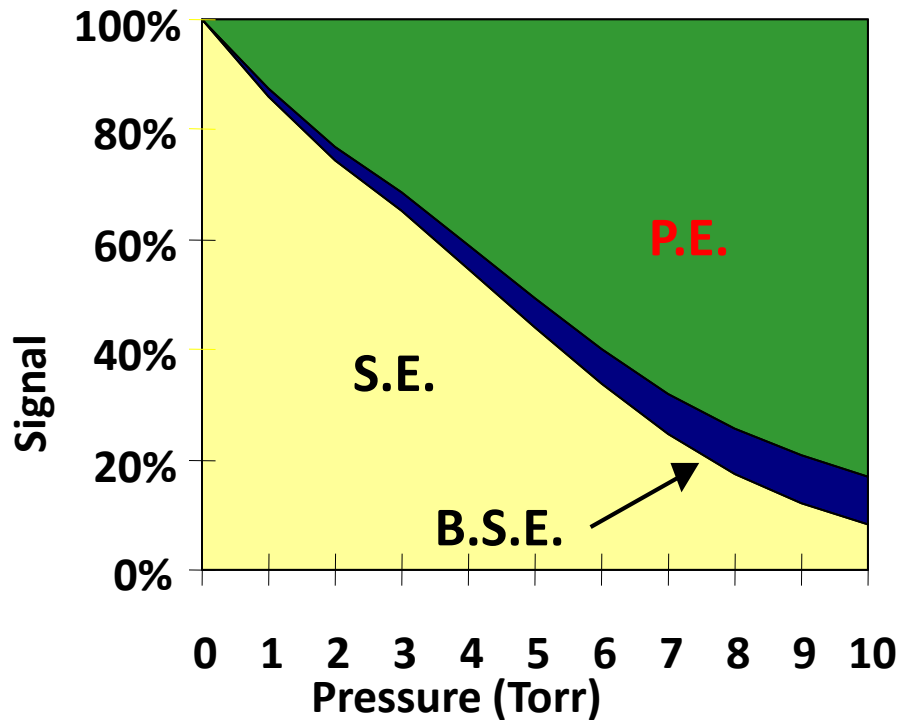
If the pressure is too small (concentration of gas molecules is small too), MFP is large and amplification events are few and the secondary electron signal is weak

If the pressure is too high (concentration of gas molecules is large), MFP is short and significant fraction of secondary electrons is scattered before they acquire sufficient energy to cause ionisation

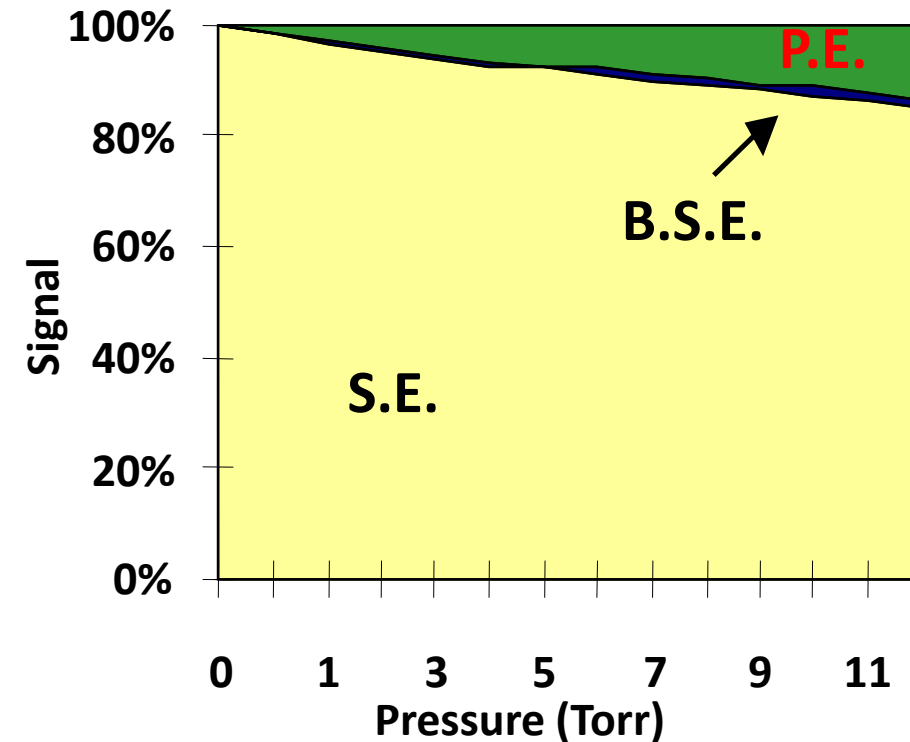


Signal Components

Water



Helium



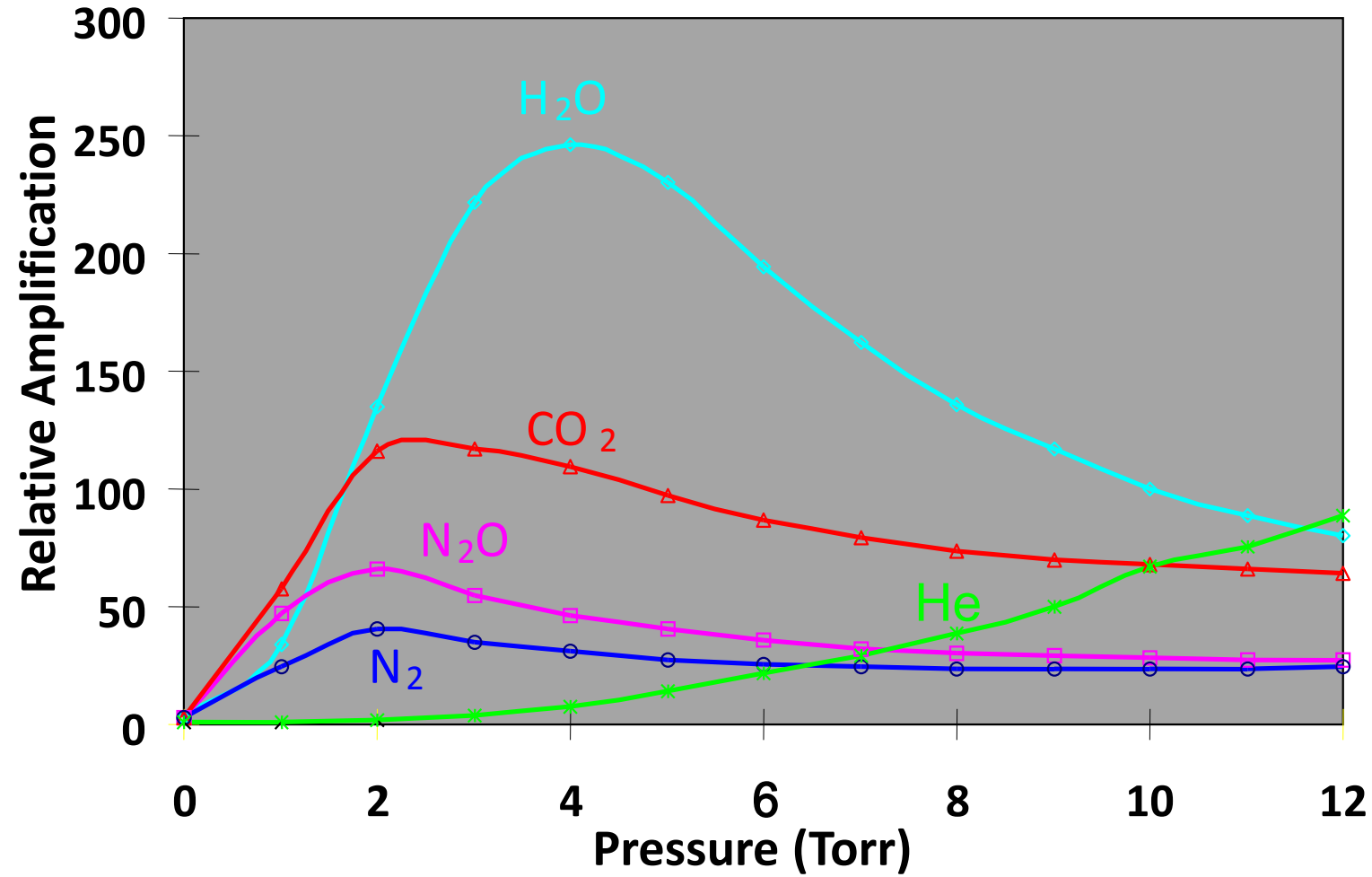
PE – signal due to ionisation from primary electrons

BSE – signal due to ionisation from backscatter electrons

SE – signal due to ionisation from secondary electrons (most desired!)



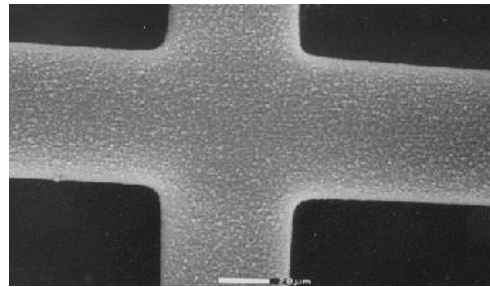
Alternative Gases



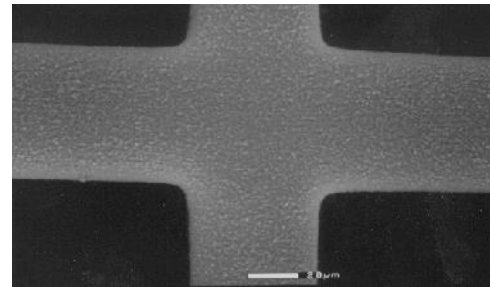


Copper grid on Carbon

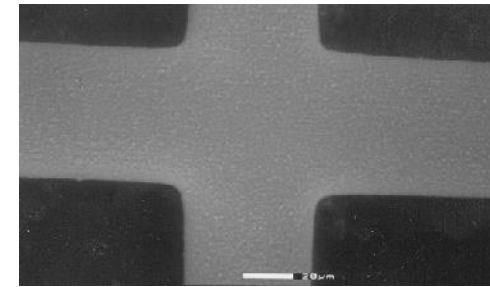
H₂O



4 torr

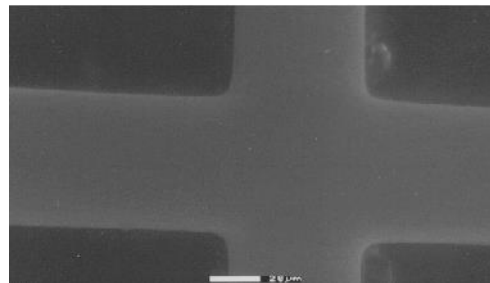


6 torr

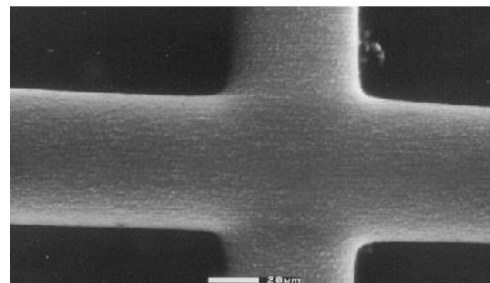


8 torr

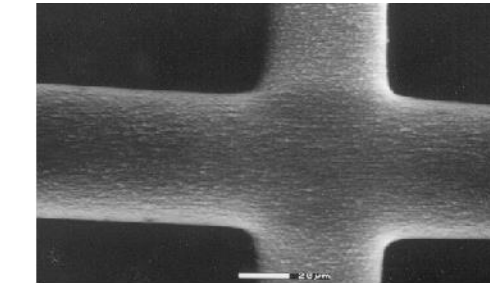
He



2 torr



5 torr



12 torr



I_c – amplified gaseous electron current

I_o – primary beam current

S_{PE} – ionisation efficiency of primary electrons

S_{BSE} – ionisation efficiency of backscatter electrons

δ – secondary electron coefficient

η – backscatter electron coefficient

α – ionisation efficiency of the gas

d – sample-anode gap

$$I_c = I_o \exp^{\alpha d} \left\{ \delta + \frac{S_{PE}}{\alpha d} + \eta \frac{S_{BSE}}{\alpha d} \right\}$$

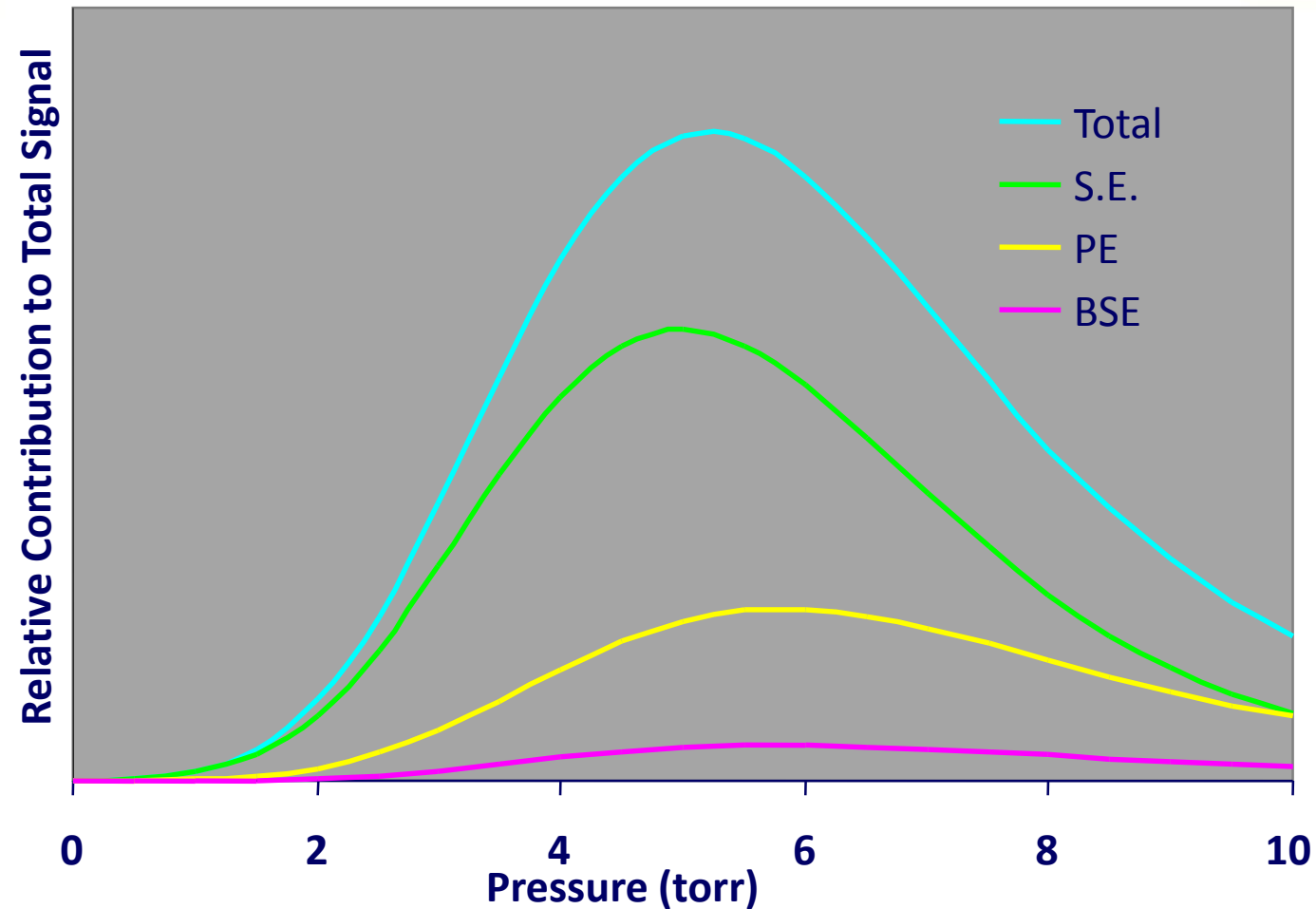
The factor α is called Townsend's first ionisation coefficient

$$\alpha = AP e^{-BPd/V_0}$$

P – pressure, V_0 – anode bias, A & B – gas specific constants

$$G = e^{\alpha d}$$

Signal gain G is a function of the ionisation coefficient α and the specimen-anode distance d



Graph of signal contributions as a function of chamber gas pressure (for water vapour). The maximum total amplification occurs at slightly higher pressure than the secondary electron maximum. At high pressures, the signal is dominated by background signals such as ionisation of gas molecules by primary electrons



Cascade Signal Components

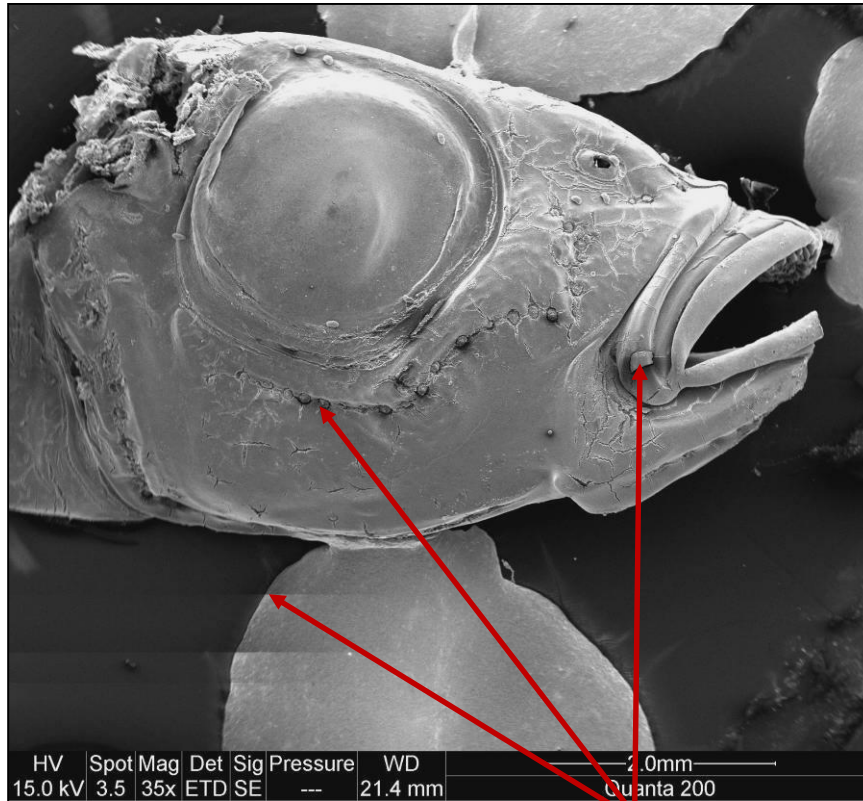
The cascade signal is then composed of:

- **Amplified SE**
- **BSE derived signal**
- **PE derived signal**
- **All of the above from the “skirt region”**
- **Detector noise**

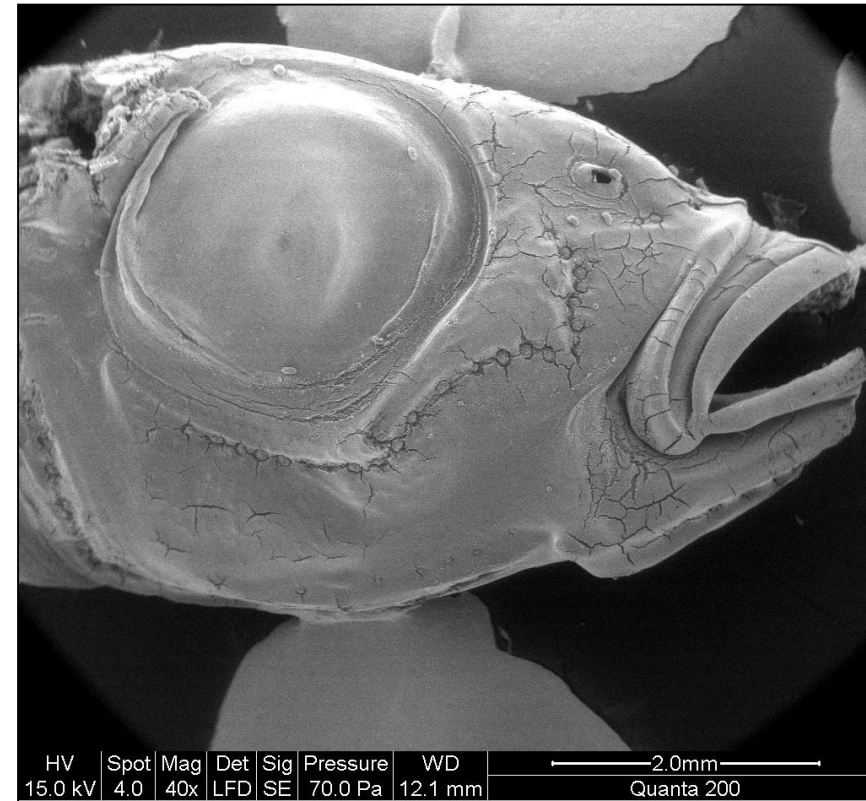
Of these, only the first is desired!



High vacuum versus low vacuum mode



SE images of Au coated Trich:
despite the coating, the image still shows
some charging effects



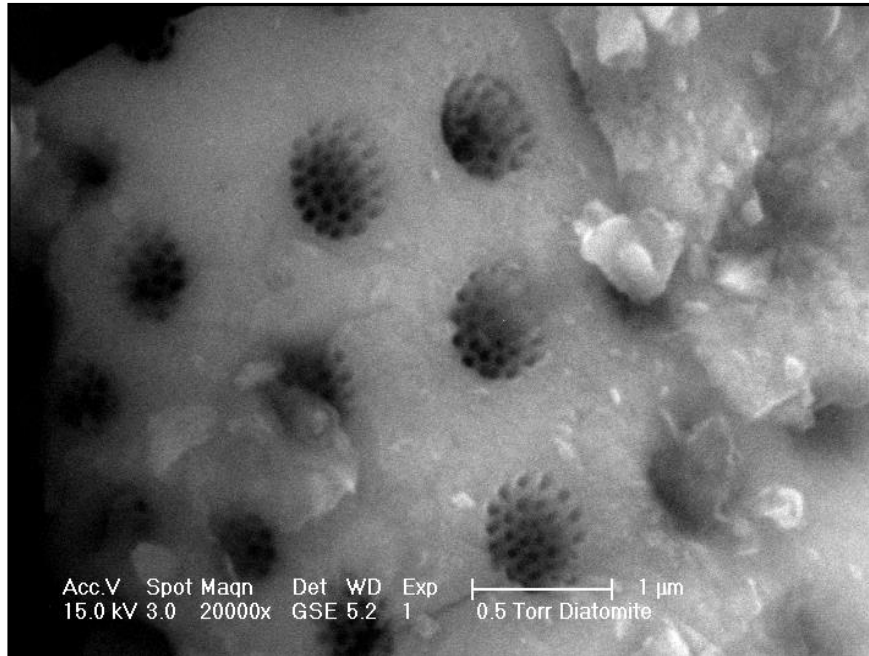
SE image of Au coated Trich made in low
vacuum mode (without charging)



Microstructure

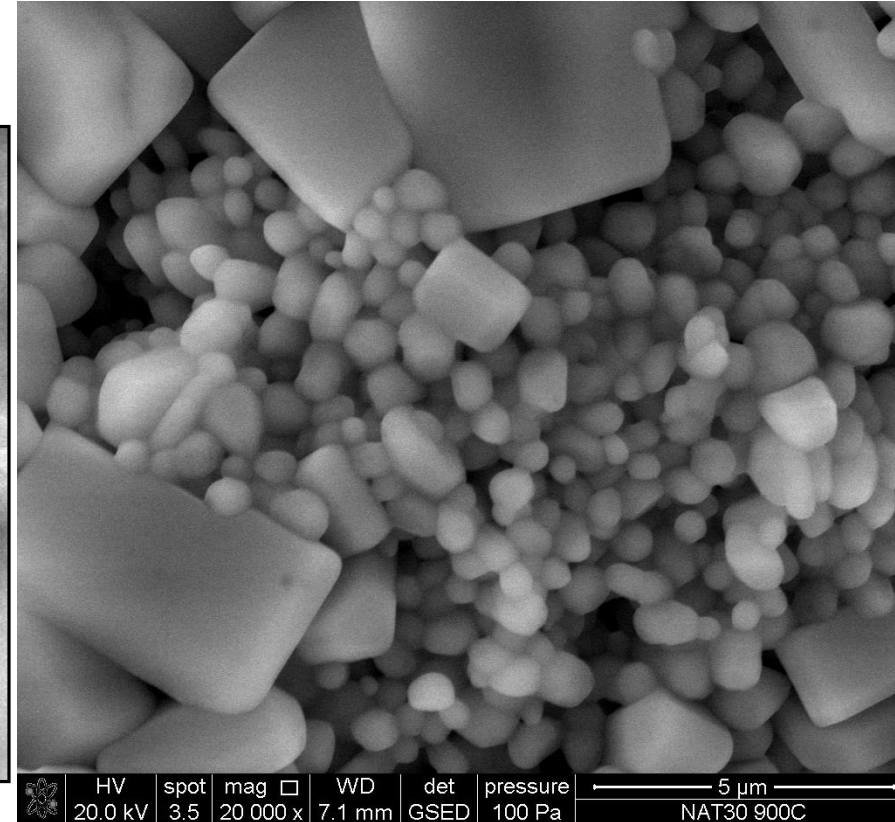
Diatomite

(a powdery mineral composed of the fossilised skeletal remains of microscopic single-celled aquatic plants called diatoms)



E-SEM with W hair-pin filament

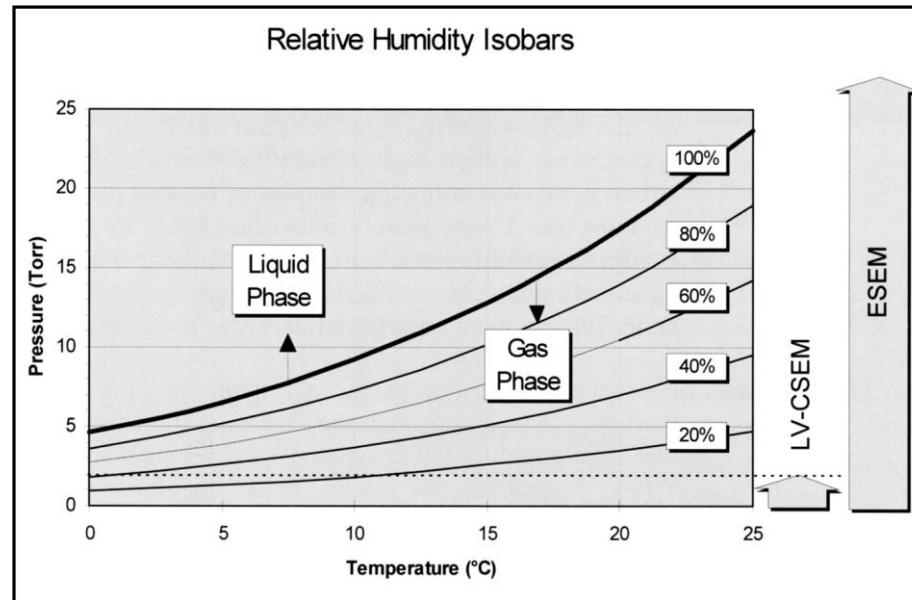
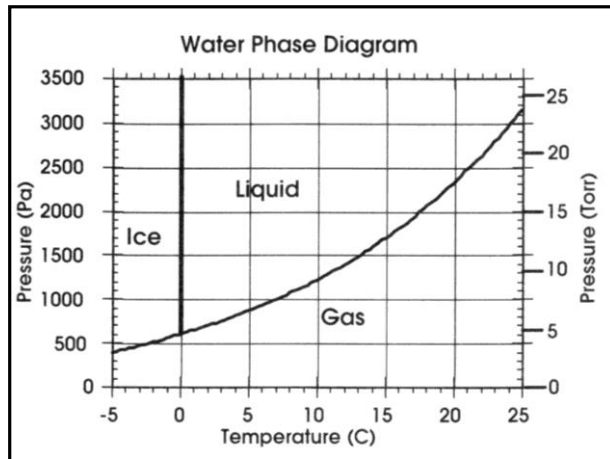
Natural HAp/bioglass composite



E-SEM with Shottky FEG



The minimum pressure that can sustain water in the liquid phase is about 4.6 Torr at 0°C. Higher temperatures require higher pressures



To maintain a sample in its natural hydrated state it is necessary to drop the temperature below room temperature,

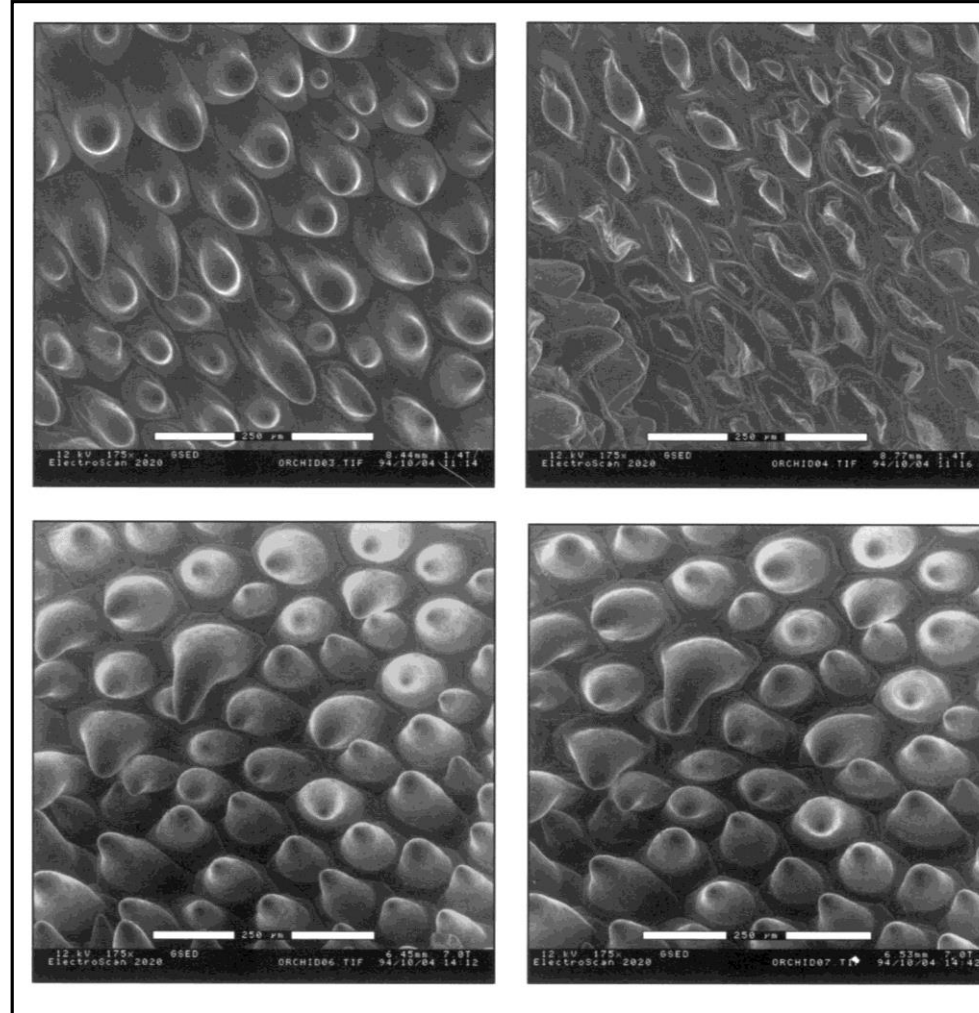
Peltier cooling stage -25°C to +50°C

before 2 min. exposure

after 2 min. exposure

1.4 Torr H₂O
vapor at 6°C
dehydration is
obvious

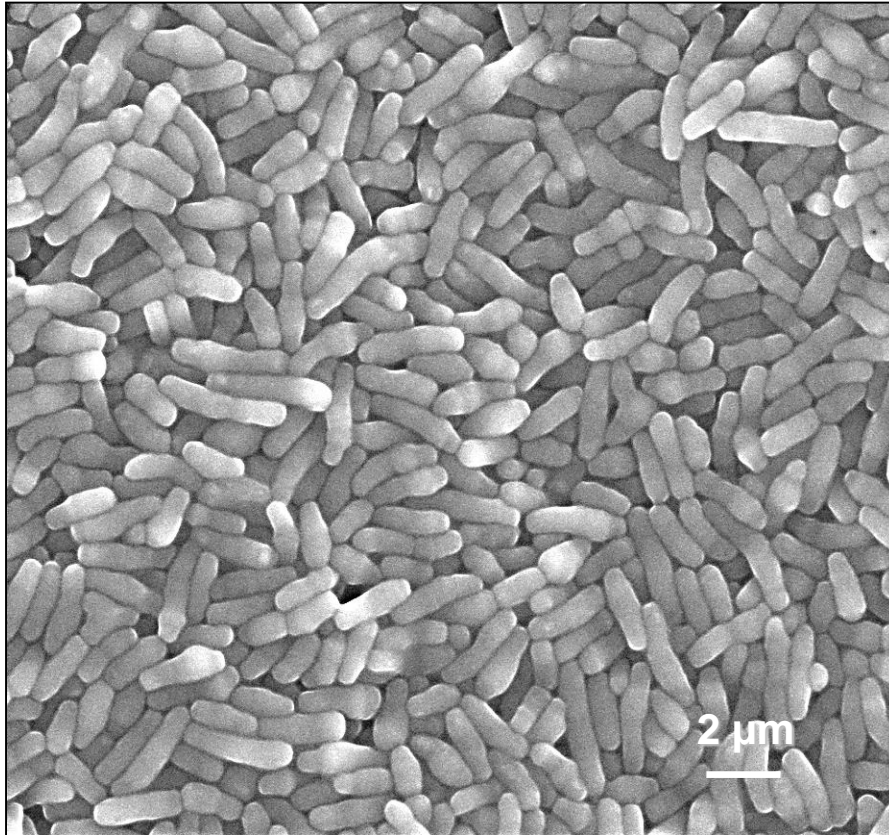
7 Torr H₂O
vapor at 6°C
H₂O environment
becomes saturated
and dehydration
does not occur



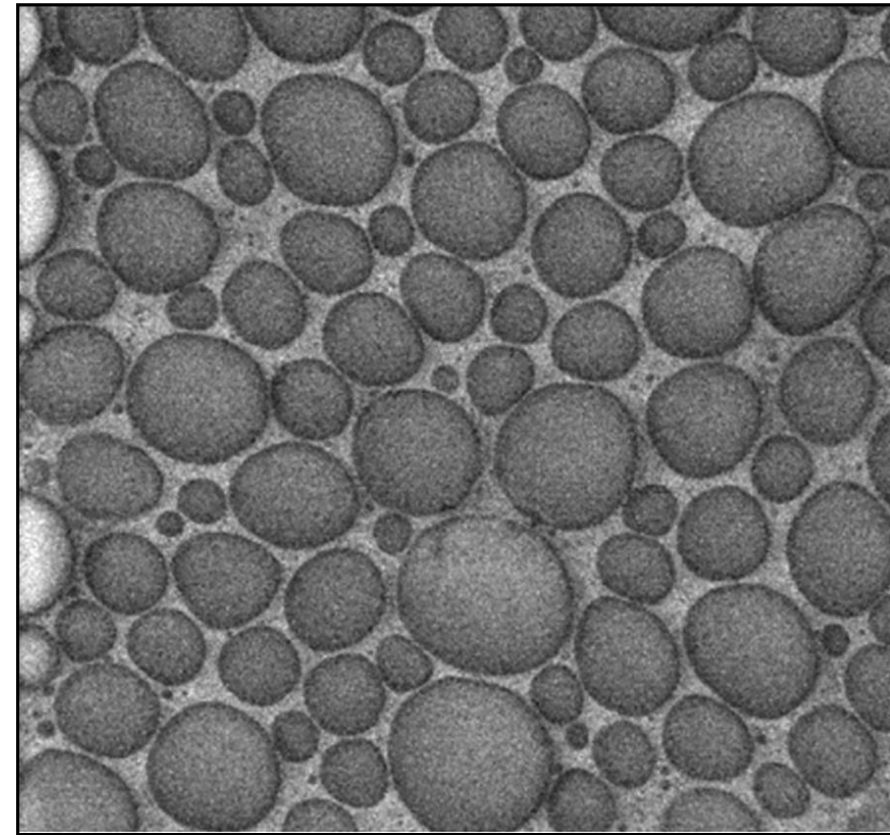
Hydrated sample
orchid petal



ESEM 'wet' mode



Fresh bacteria



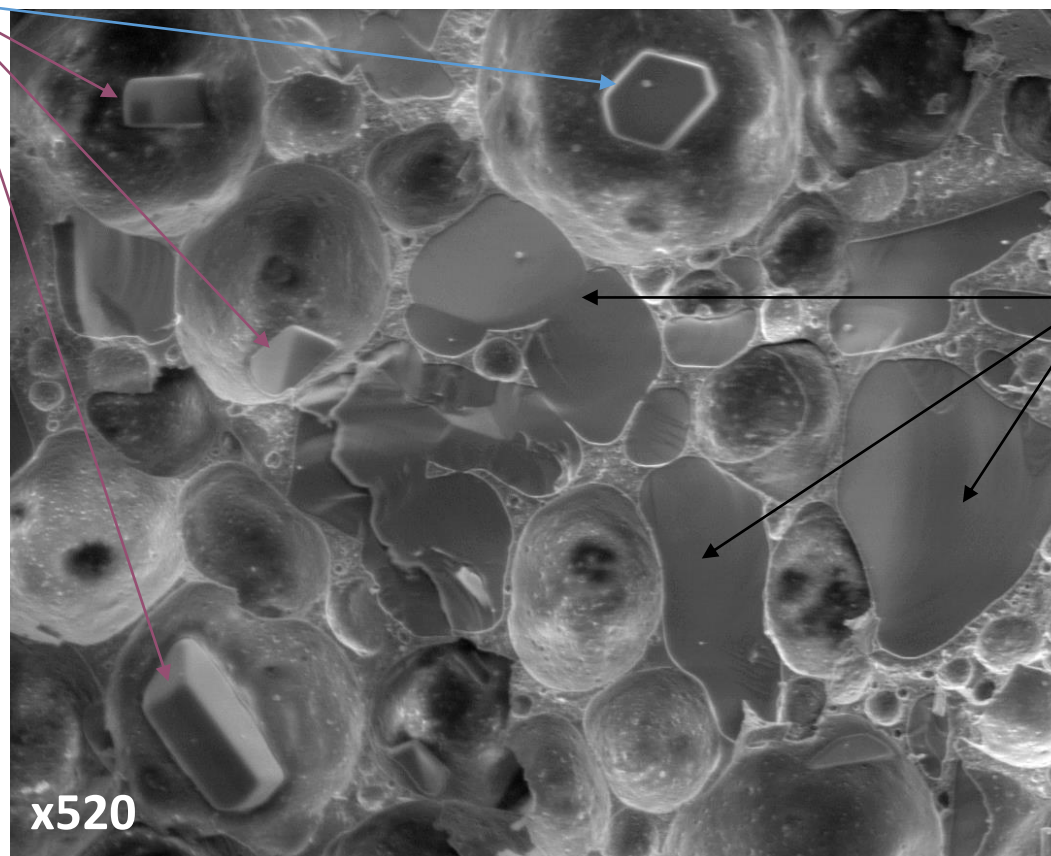
Oil-in-water emulsion



Interfacial ice

$T = -120^{\circ}\text{C}$ $P = 0.7 \text{ Torr N}_2$

**Matrix
ice**



Stokes, Mugnier & Clarke (2004) J Microscopy 213(Pt 3)





Microanalysis under low vacuum conditions

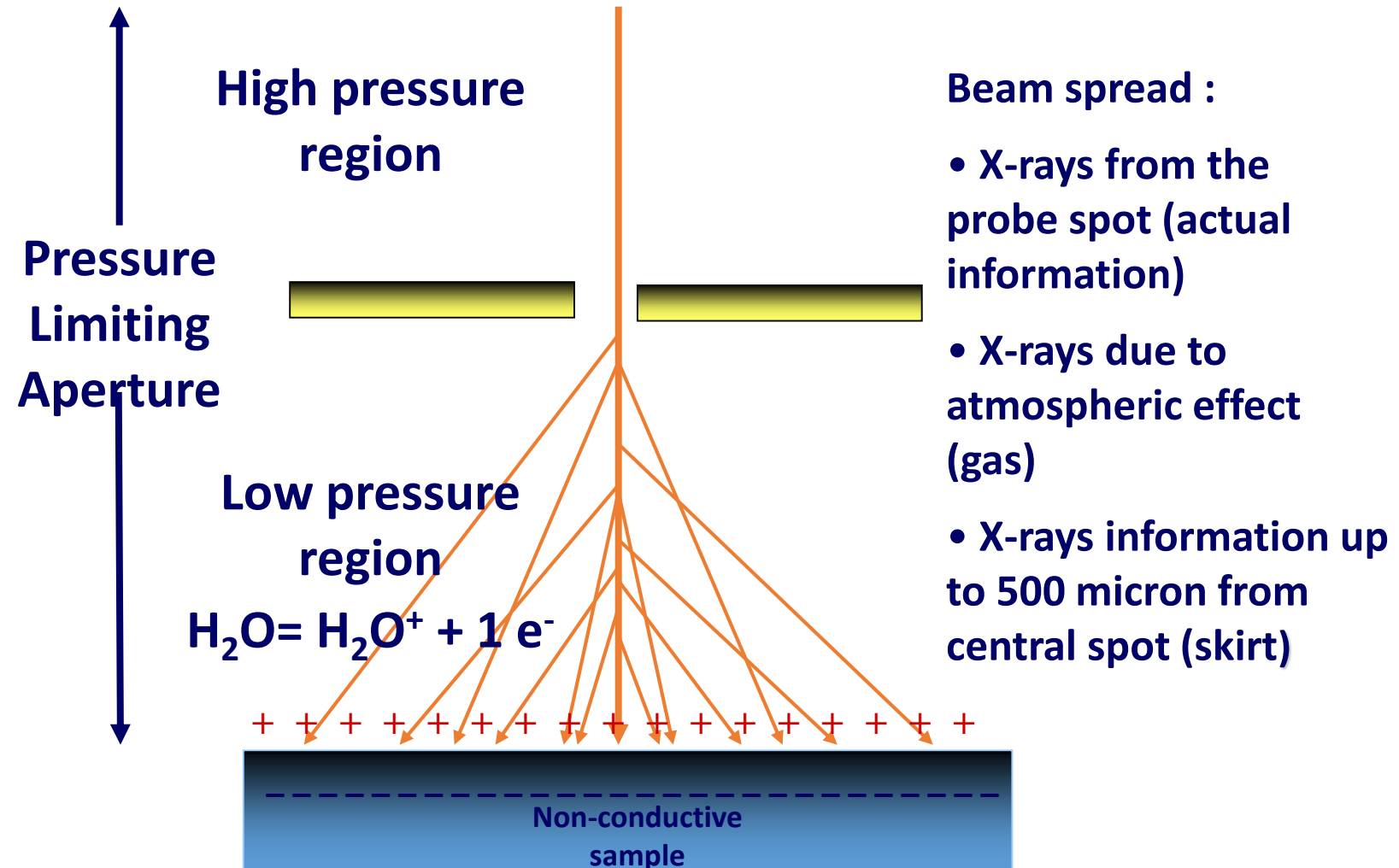
- **Low vacuum SEM: charge is eliminated by a gas (water, air or N₂)**
- **Two major problems :**
 - **beam damage (heating of sample)**
 - **beam spread (skirt effect)**

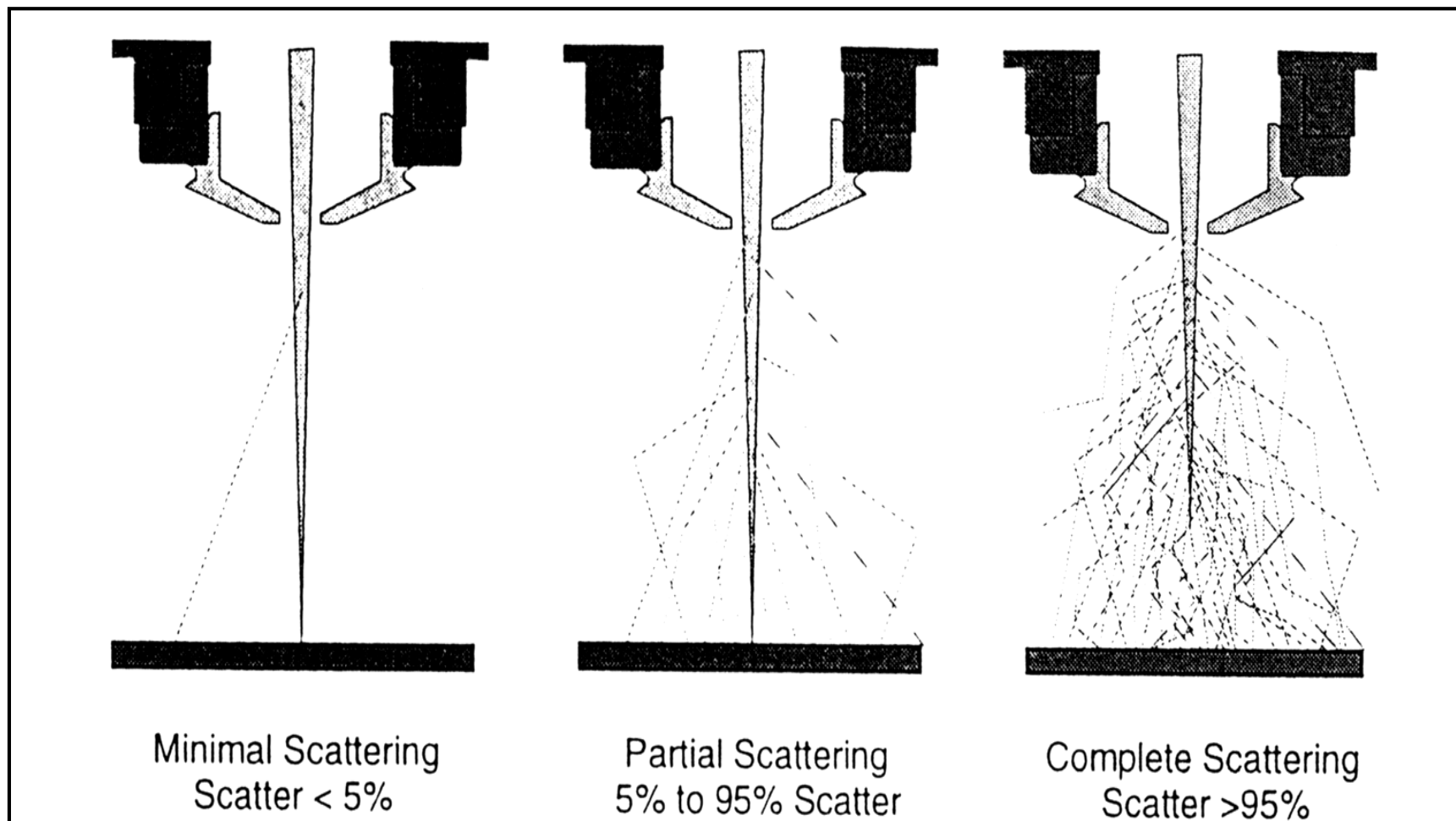
Beam spread

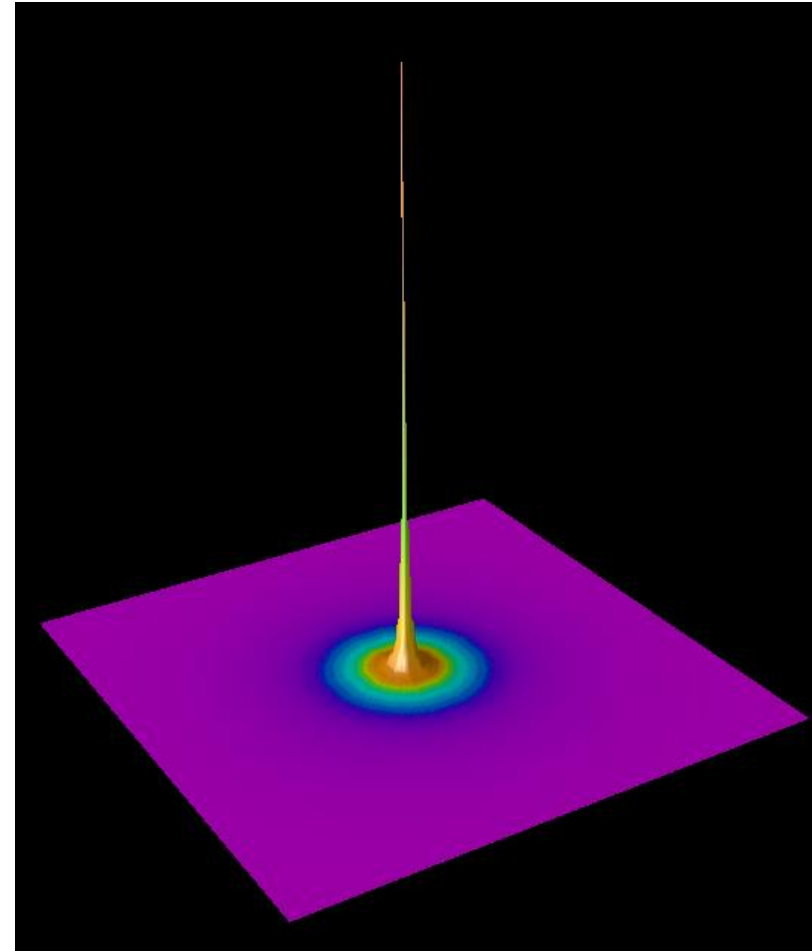
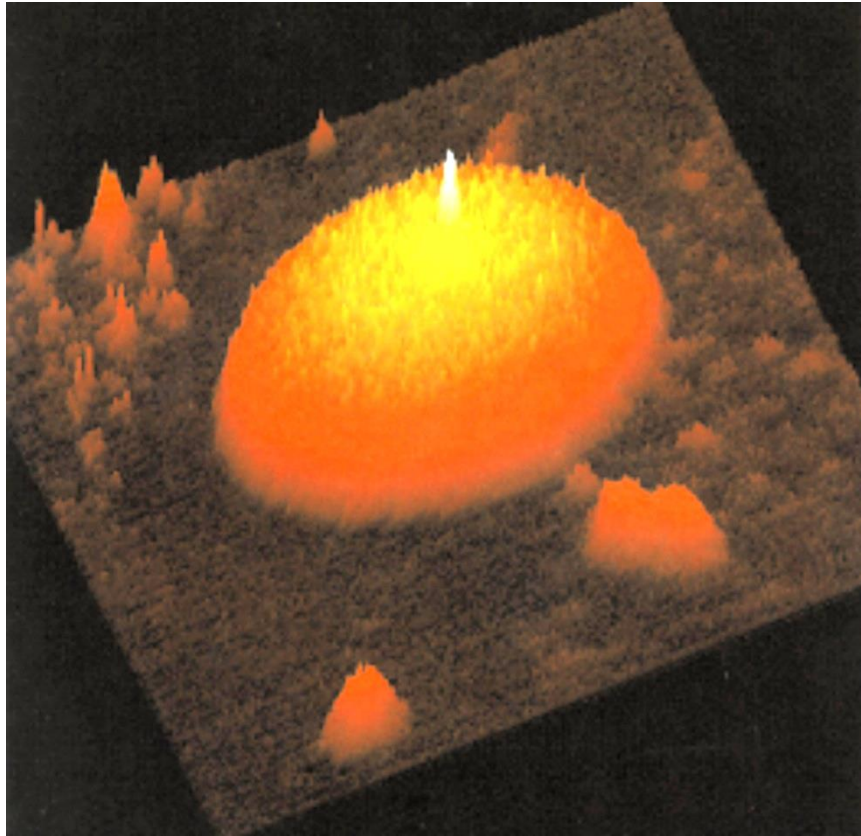
- **Electrons are scattered due to collisions with gas atoms**
- **X-ray is generated outside the probe**
- **X-ray information comes from up to 500 micron from central spot (skirt area)**



Elastic scattering – Skirt formation







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Electrons in the Skirt

Fraction of electrons remaining in central probe:

$$f = e^{-\sigma P d / RT}$$

where:

σ – scattering cross section

P – pressure

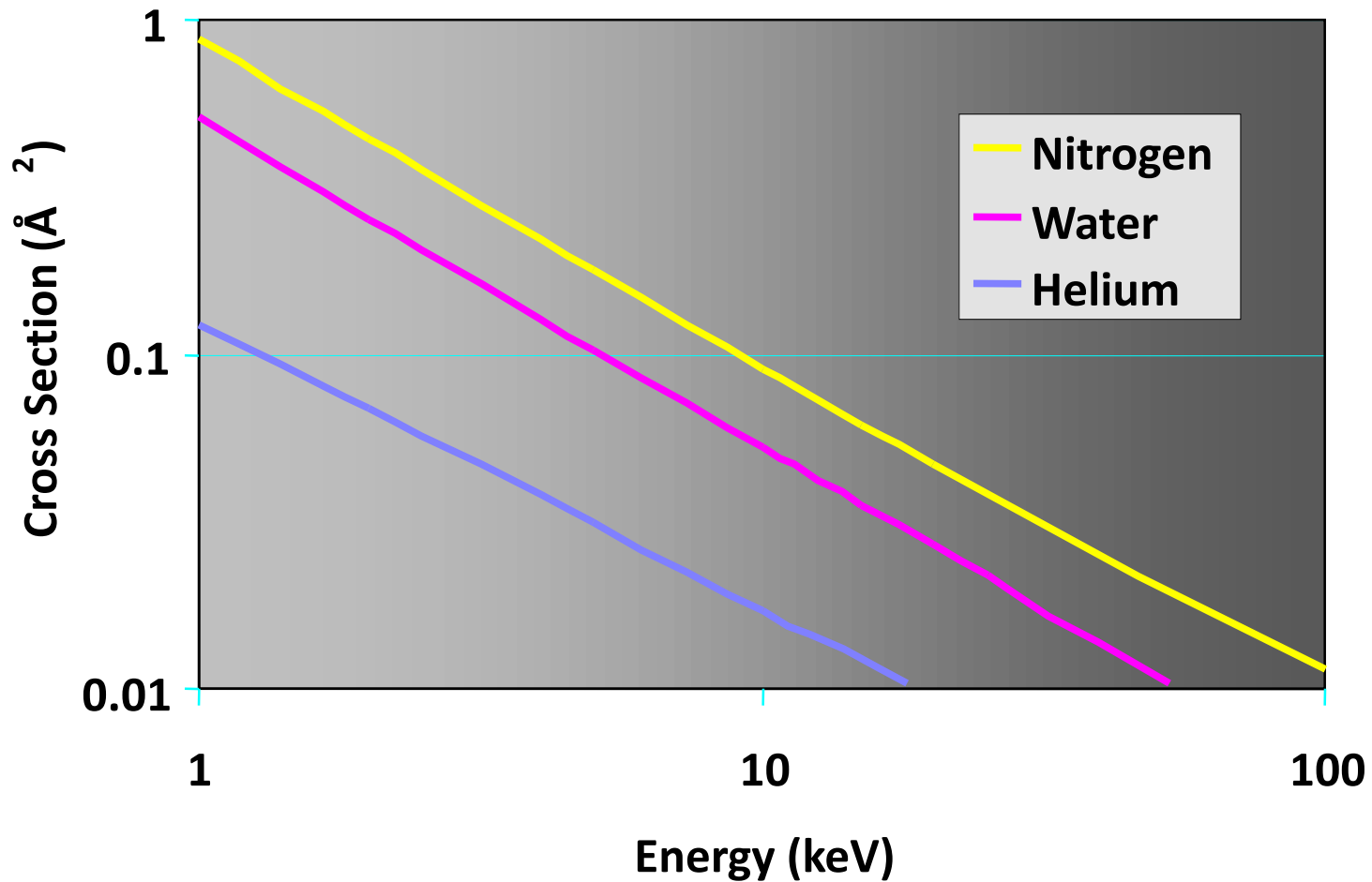
R – gas constant

d – gas path length

T – temperature

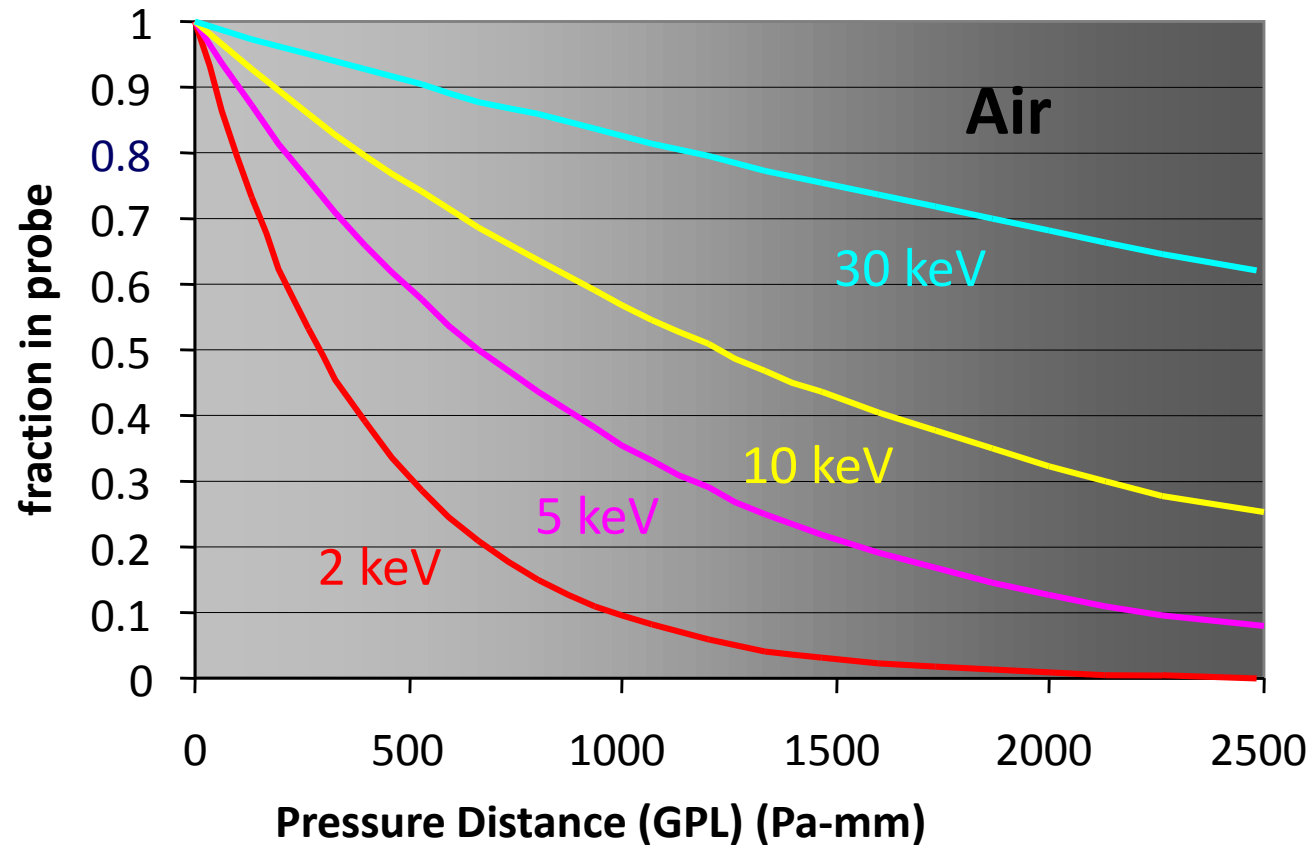


Elastic Cross Sections





Scattering out of the Probe



Use as low a pressure and as short a Gas Path Length as possible!



Which gas?

Influence of atomic number on primary electron mean free path λ as a function of primary beam energy

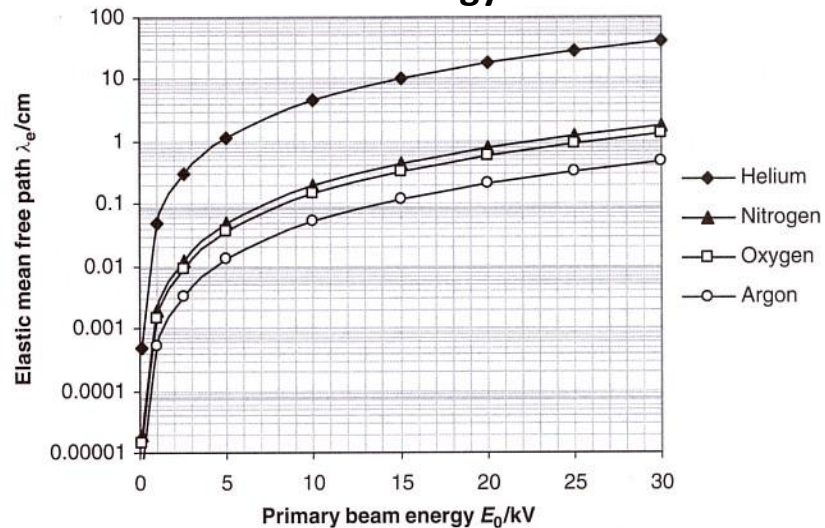


Figure 4.2 Log-linear plot of primary electron mean free paths as a function of atomic number for a range of primary beam energies. The data points are for helium ($Z = 2$), nitrogen ($Z = 7$), oxygen ($Z = 8$) and argon ($Z = 18$). Pressure $p = 100$ Pa

Influence of atomic number on skirt radius r_s as a function of primary beam energy

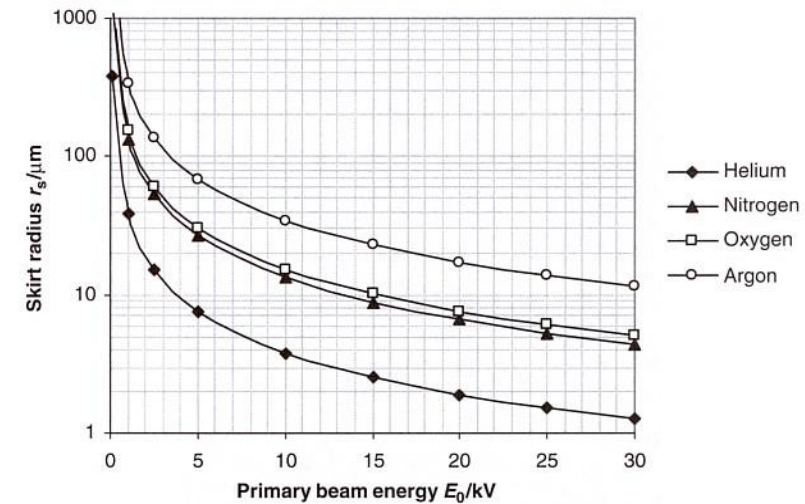


Figure 4.3 Log-linear plot of skirt radius r_s as a function of primary beam energy for a range of gases having atomic numbers $Z = 2$ (helium), $Z = 7$ (nitrogen), $Z = 8$ (oxygen) and $Z = 18$ (argon). Thickness of the gas layer (gas path length) = 2 mm. The temperature is assumed to be $T = 293$ K (20°C) and pressure $p = 100$ Pa (0.75 torr)



**Helium –
theoretically the
best gas.
However it is the
most difficult gas to
ionize and the small
size of helium
atoms makes this
gas notoriously
difficult to handle
with a typical
vacuum pump.
Prolonged use of
helium is definitely
not recommended.**

Influence of atomic number on the percentage of primary electrons remaining in the focused probe as a function of primary beam energy

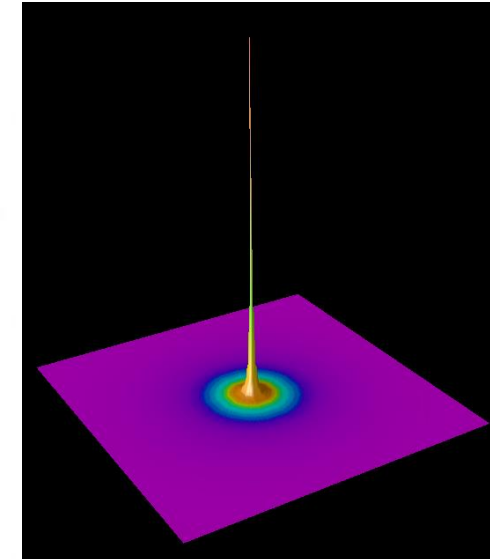
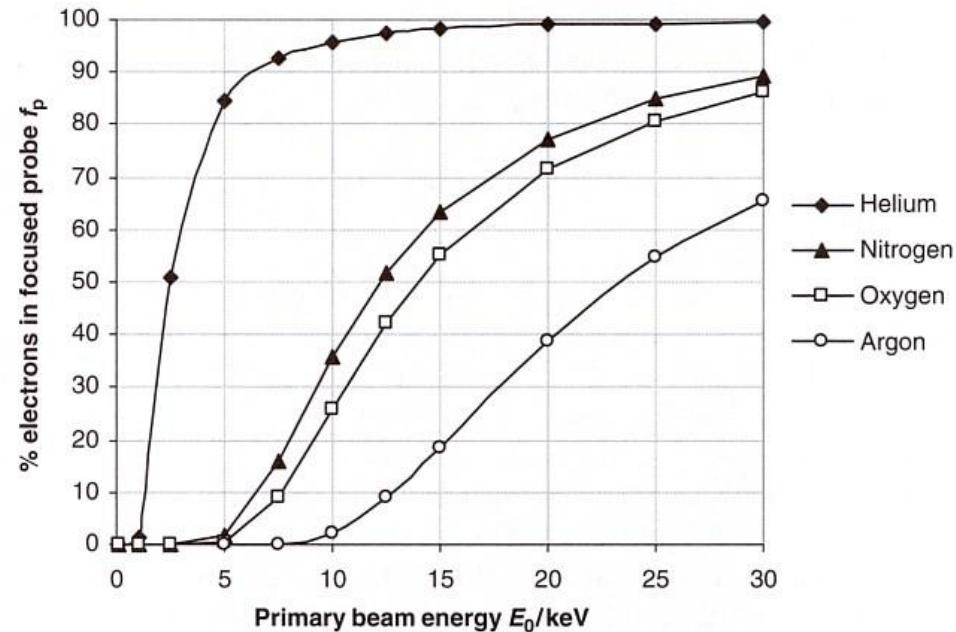
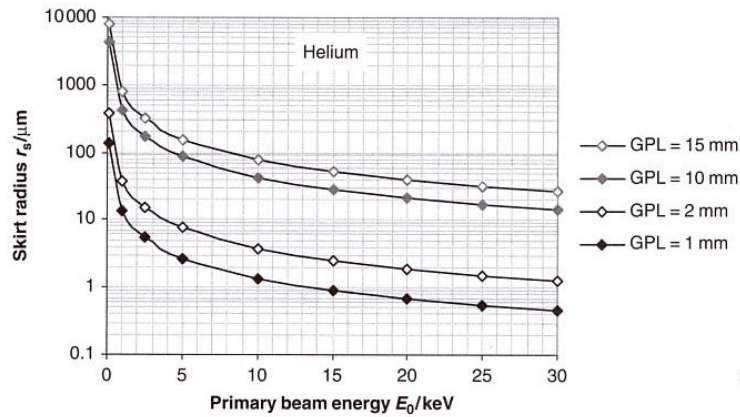
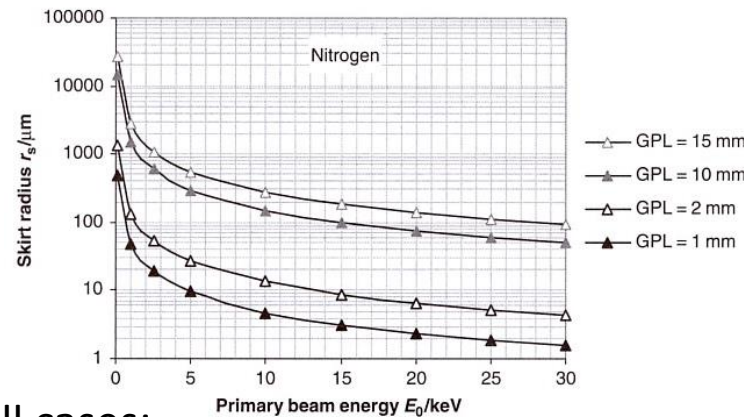


Figure 4.5 Plot of the percentage of primary electrons remaining in the focused probe to form a useful beam current, as a function of primary electron beam energy E_0 for a range of gases with atomic numbers $Z = 2$ (helium), $Z = 7$ (nitrogen), $Z = 8$ (oxygen) and $Z = 18$ (argon). Thickness of gas layer (gas path length) = 2 mm, pressure $p = 100$ Pa



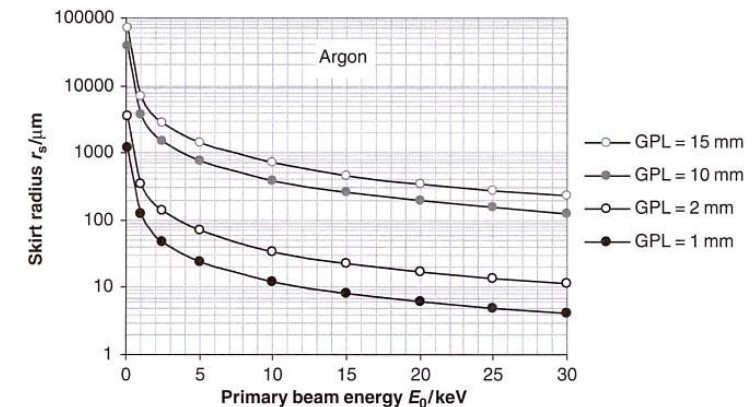
Helium

Exploring the Gas Path Length



Nitrogen

The general trends can be seen in all cases:
there is a sharp rise in the radius of the skirt as the primary beam energy decreases below about 5 keV and almost two orders of magnitude difference between radii for GPL = 1 mm compared to 15 = mm



Argon



With the exception of helium, the percentage of electrons remaining in the focused probe falls sharply as the GPL increases. That brings with it a reduction in electron beam current. For argon, it would be advisable to maintain gas path length of a millimeter or two. For nitrogen and oxygen the GPL of the order a few millimeters is advisable.

Exploring the Gas Path Length

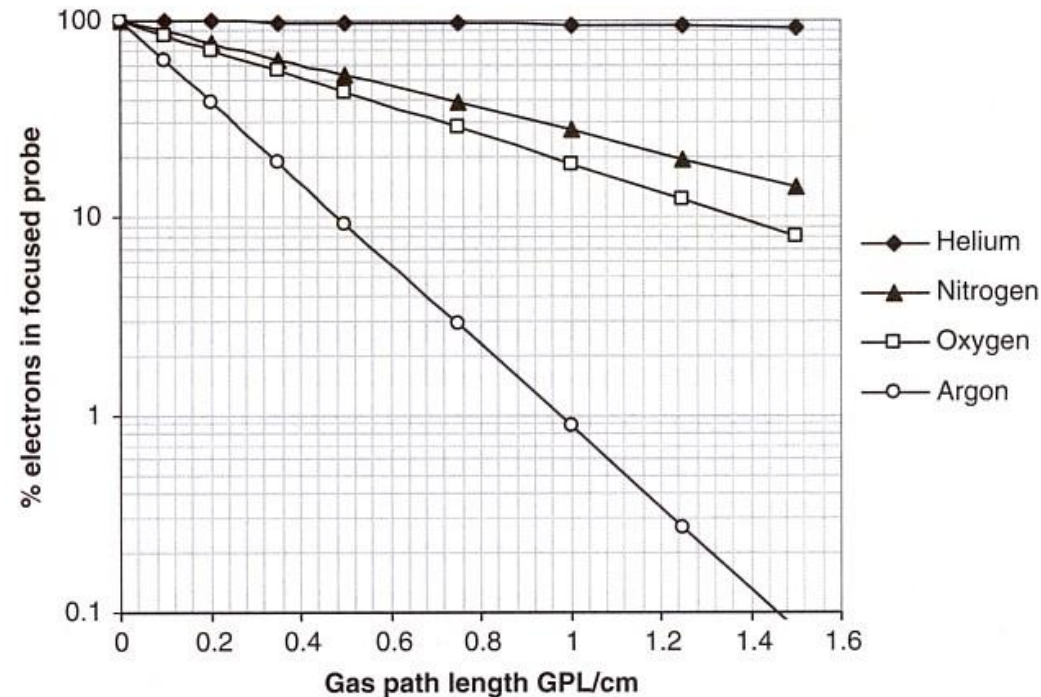


Figure 4.9 Log-linear plot to show the percentage of electrons remaining in the focused probe to form the useful primary current, as a function of atomic number of the gas for a primary beam energy $E_0 = 20 \text{ keV}$. Pressure $p = 100 \text{ Pa}$

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How much gas?

Influence of energy on elastic mean free path λ as a function of pressure for nitrogen

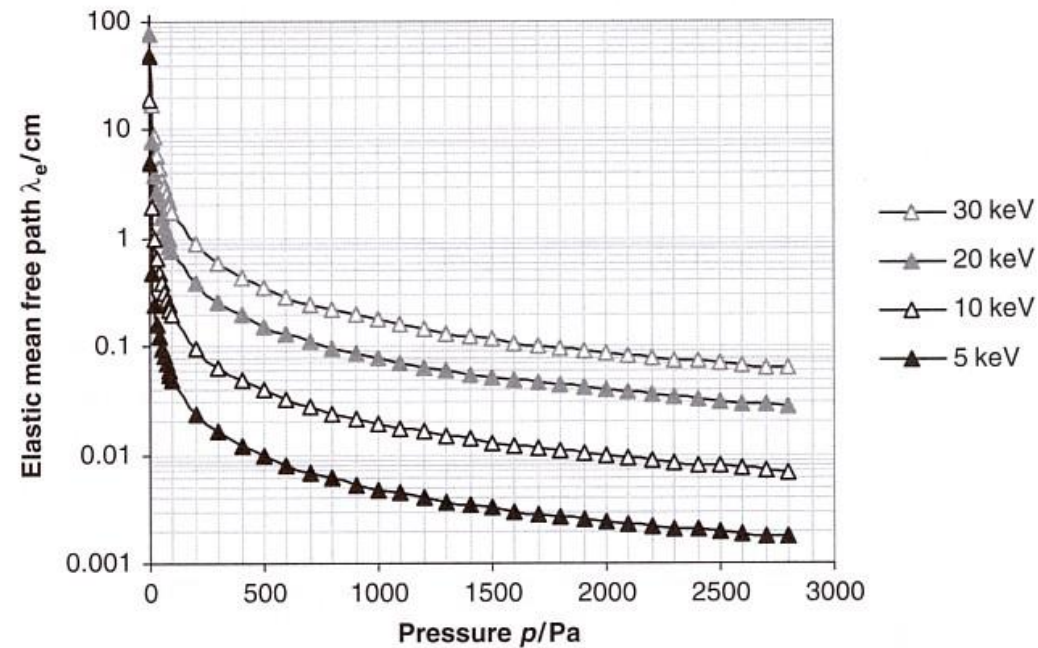


Figure 4.14 Log-linear plot to show the variation in elastic mean free path λ_e for primary electrons in nitrogen gas over the wide range of pressures available in the VP-ESEM and for several primary beam energies E_0



How much gas?

Influence of atomic number on skirt radius r_s as a function of pressure

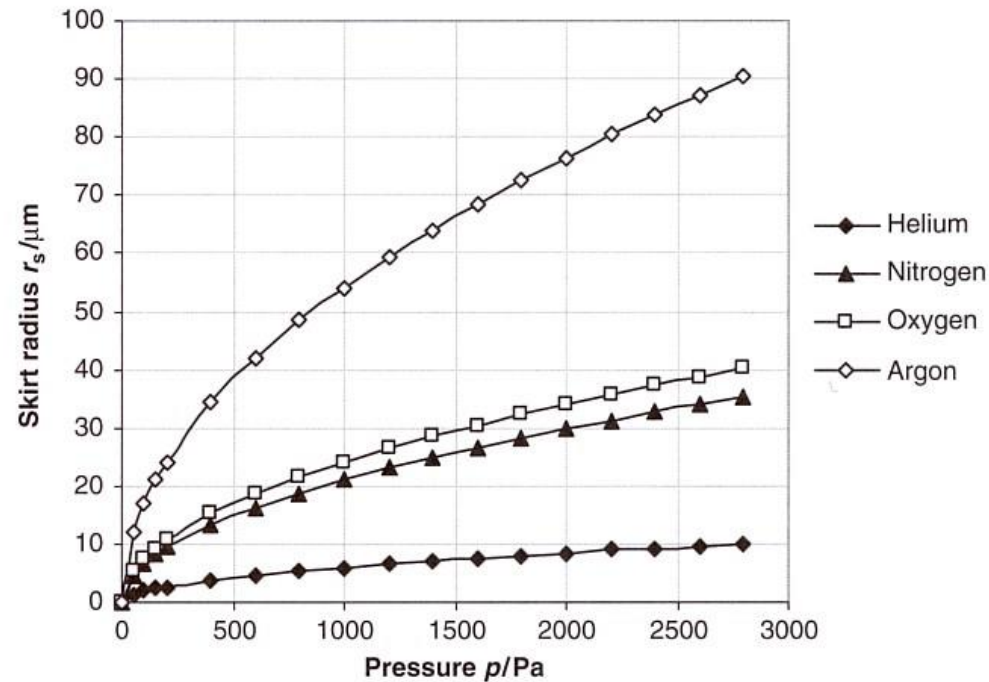
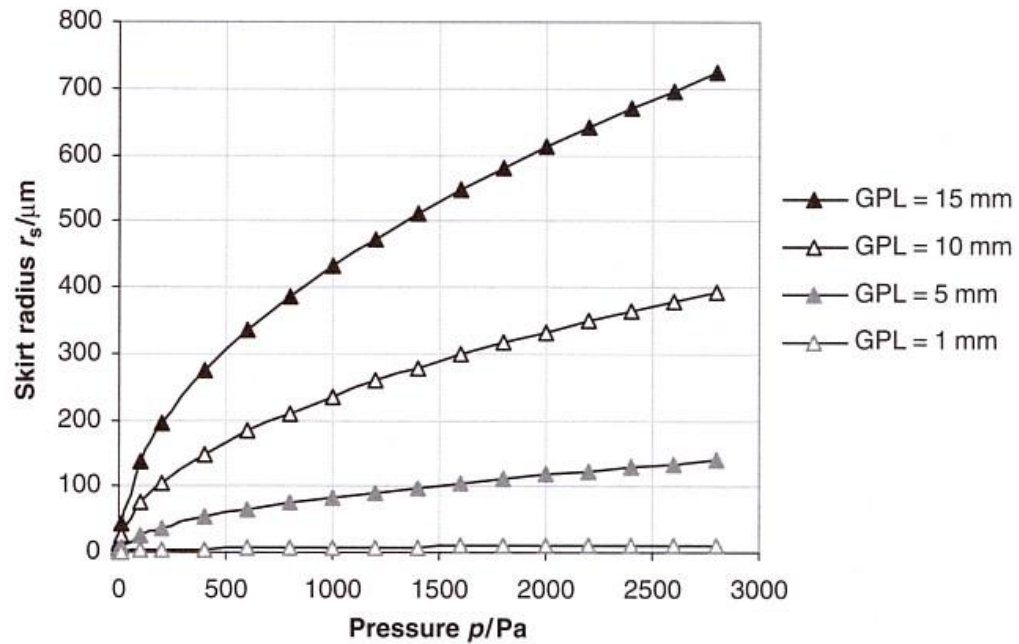


Figure 4.16 Plot of skirt radii r_s over the pressure range extending to 2.8 kPa for gases having atomic numbers $Z = 2$ (helium), $Z = 7$ (nitrogen), $Z = 8$ (oxygen) and $Z = 18$ (argon). Primary beam energy $E_0 = 20$ keV and gas path length GPL = 2 mm



How much gas?

Influence of pressure on skirt radius r_s for several GPLs for nitrogen



$p = 100 \text{ Pa}$

$R_s = 2.4 \text{ } \mu\text{m}$ for GPL = 1 mm

$r_s = 26 \text{ } \mu\text{m}$ for GPL = 5 mm

$p = 2.8 \text{ kPa}$

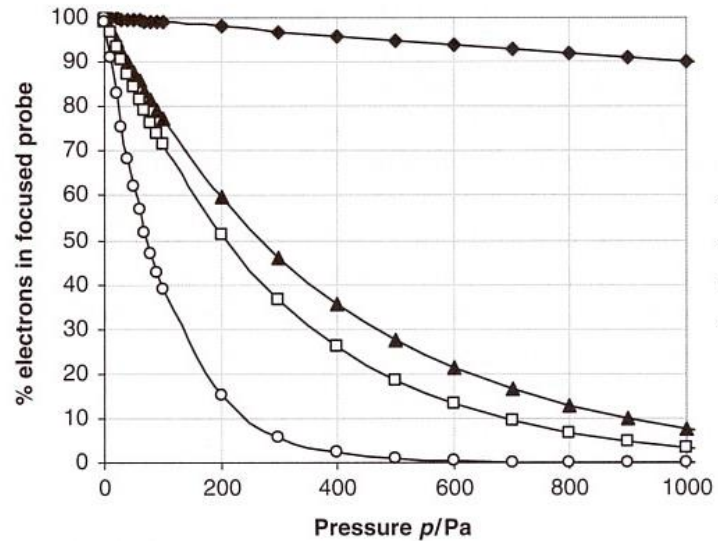
$r_s = 12.5 \text{ } \mu\text{m}$ for GPL = 1 mm

$r_s = 139 \text{ } \mu\text{m}$ for GPL = 5 mm

Figure 4.17 Plot of primary beam skirt radii r_s over the pressure range extending to 2.8 kPa for several gas path lengths and in nitrogen gas, $Z = 7$. Primary beam energy $E_0 = 20 \text{ keV}$

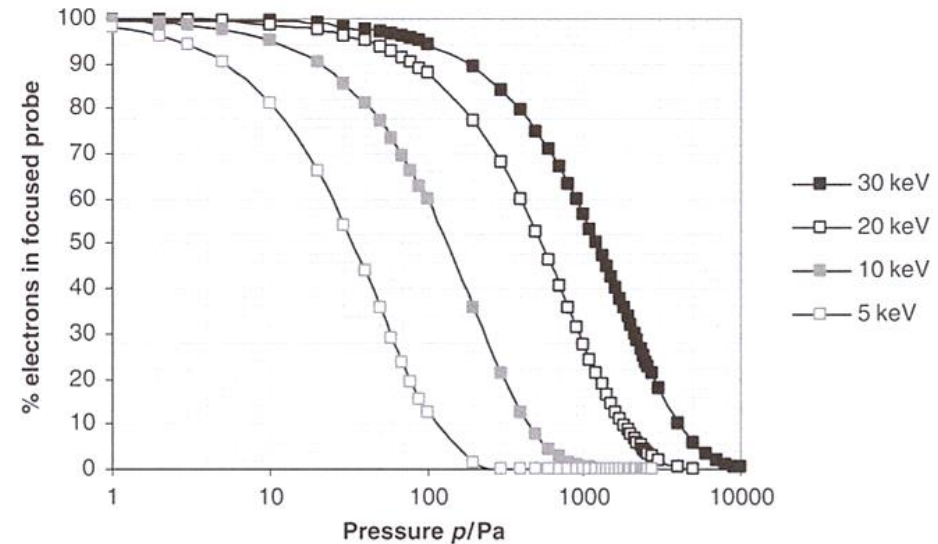


How much gas?



Percentage of electron remaining in the focused probe as a function of pressure and atomic number

GPL = 2 mm, $E_0 = 20$ keV



Percentage of electron remaining in the focused probe as a function of pressure and a range of primary beam energy

GPL = 1 mm, nitrogen

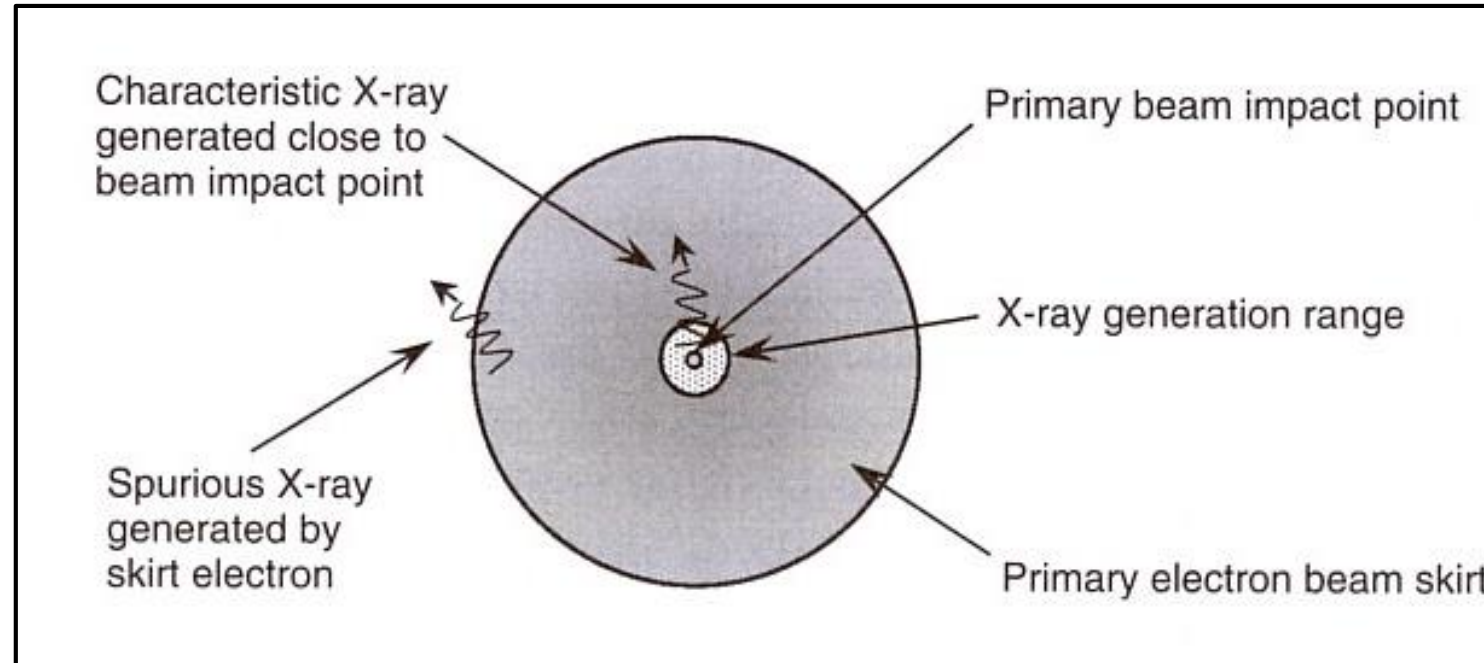


Microanalysis under low vacuum conditions

- **Low vacuum SEM: charge is eliminated by a gas (water, air or N₂)**
- **Two major problems :**
 - **beam damage (heating of sample)**
 - **beam spread (skirt effect)**

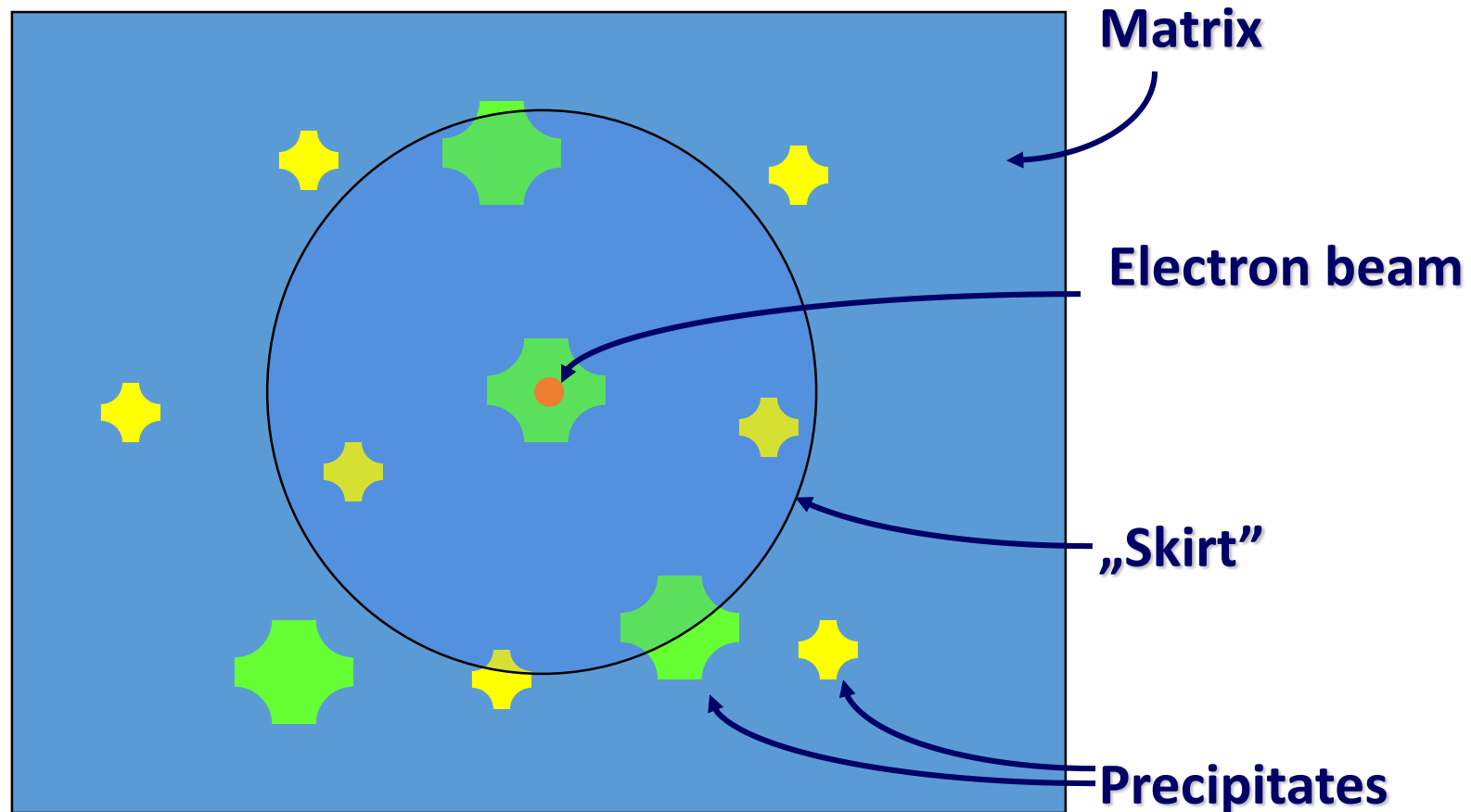
Beam spread

- **Electrons are scattered due to collisions with gas atoms**
- **X-ray is generated outside the probe**
- **X-ray information comes from up to 500 micron from central spot (skirt area)**





X-ray Skirt





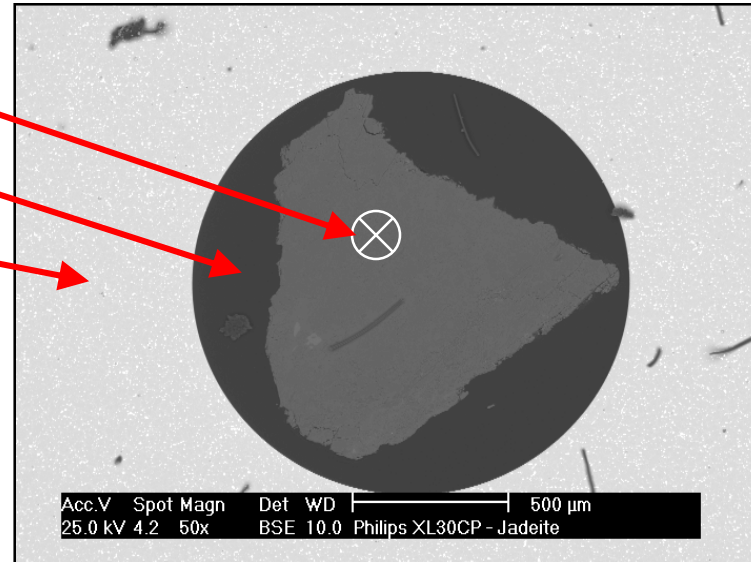
Large sample in homogeneous matrix

Sample : Jadeite $\text{NaAlSi}_2\text{O}_6$

Matrix 1 : polymer

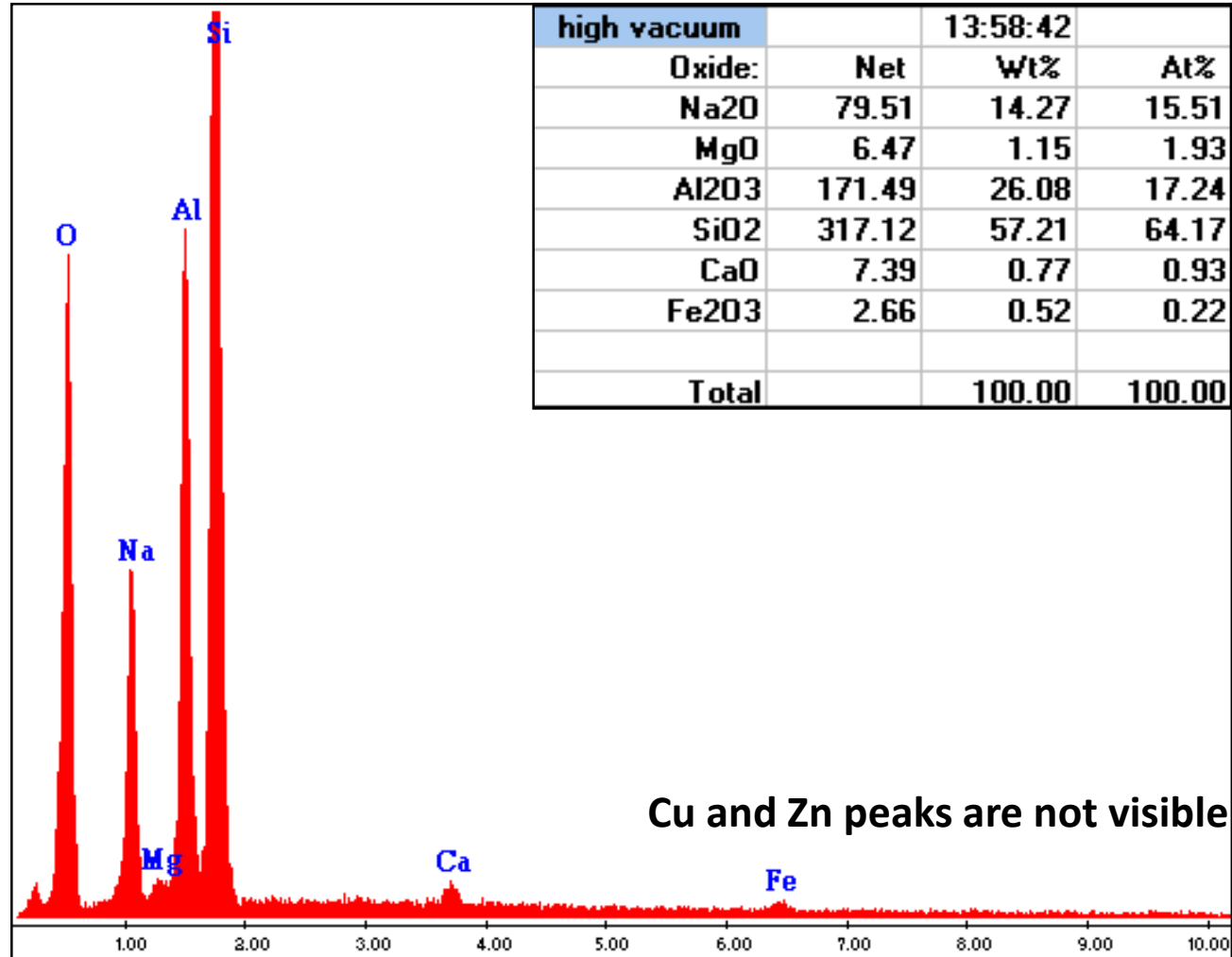
Matrix 2 : Cu/Zn block

Distance between the spot
position (centre of mineral)
and the Cu/Zn block is
more than 500 micron





High-vacuum spectrum - Jadeite 25kV





Microanalysis under low vacuum conditions

VP gas compensated technique
(with 2 spectra measured at different pressures)
developed by Wernisch and Dijkstra

To reduce the skirt effect:

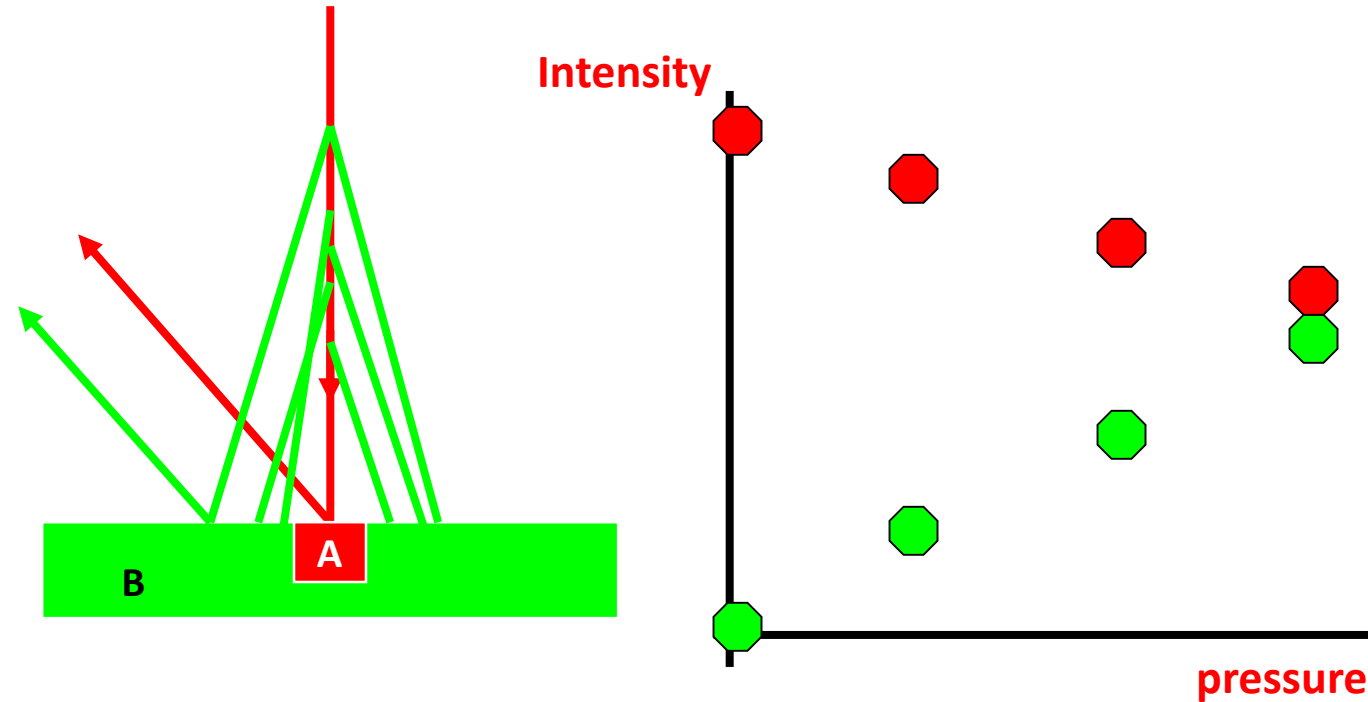
- use short gas path (a special cone attached to pole piece)
- use high acceleration voltage (25kV)
- use low pressure (0.1 - 0.3 mbar)



A pressure variation method
involves collecting spectra at two different pressures (with other parameters constant) and uses the difference spectrum to extrapolate back to zero-scattering case.

**This method assumes that the composition does not change with pressure!!!
It is only X-ray intensity that changes!**

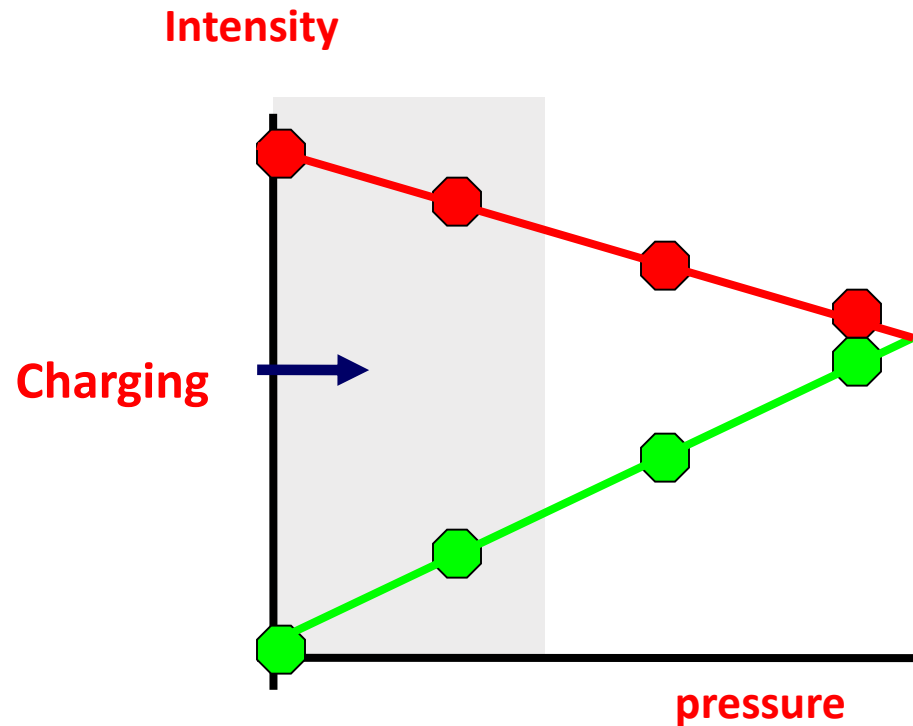
Pressure variation method



More pressure: more X-rays from B, less from A



Pressure variation method

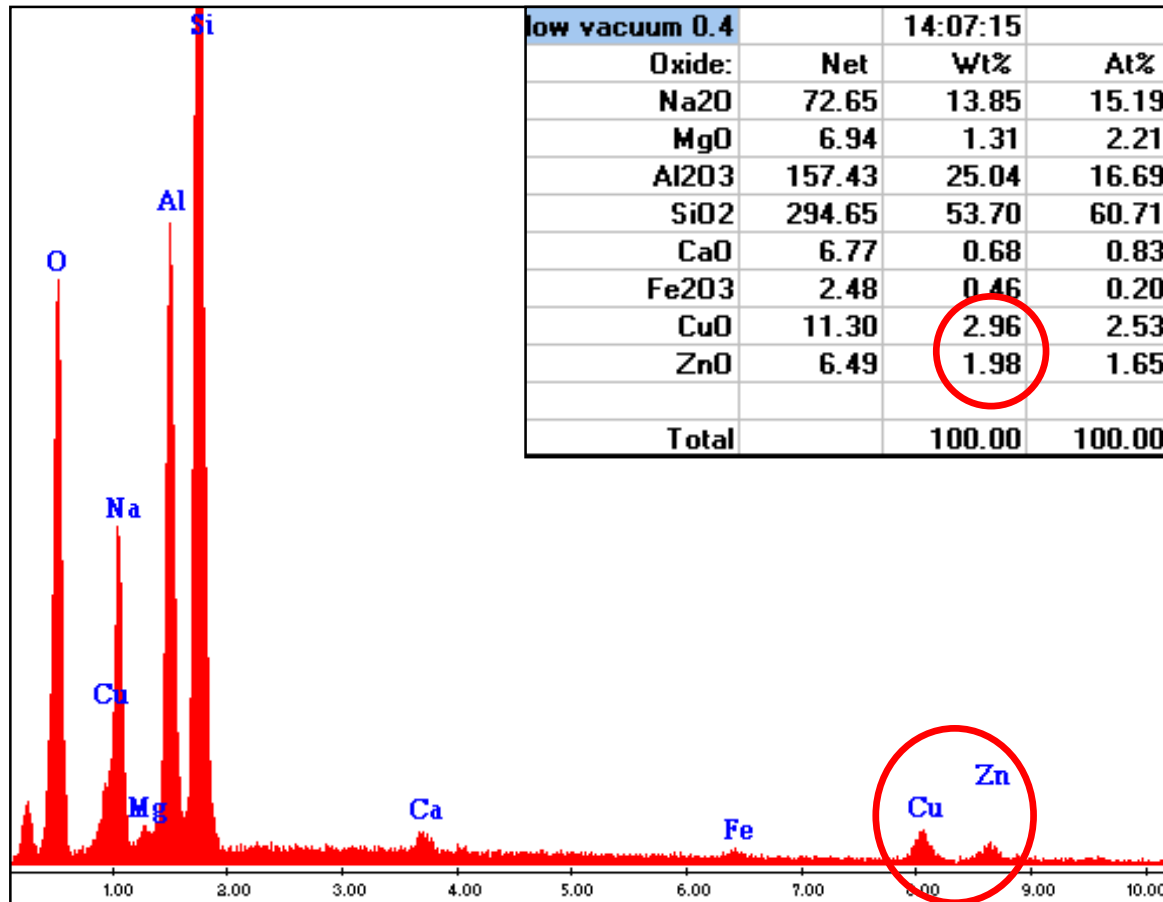


Take 2 measurements
at 2 different pressures

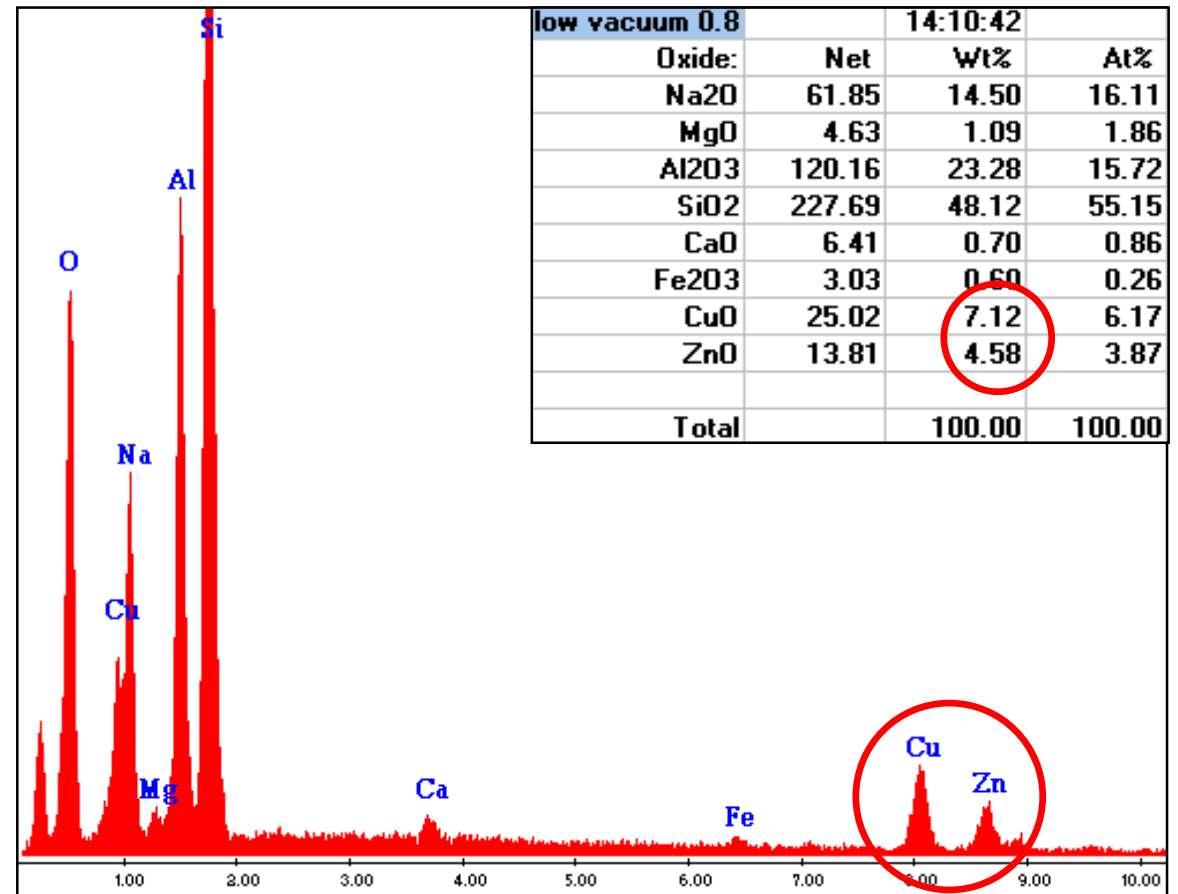
Extrapolate the intensities
to pressure 0

Apply matrix correction to
extrapolated intensities
to get composition

Low-vacuum spectrum 0.4 mbar 25kV



Low-vacuum spectrum 0.8 mbar 25kV



Contribution of the matrix (brass holder) is clearly visible

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Results

Low Vac
(0.2 - 0.4 mbar)

corrected		14:08:31	
Oxide:	Net	Wt%	At%
Na2O	78.35	14.33	15.58
MgO	7.32	1.33	2.23
Al2O3	168.07	26.12	17.27
SiO2	308.47	56.83	63.76
CaO	6.99	0.74	0.89
Fe2O3	3.18	0.64	0.27
CuO	0.00	0.00	0.00
ZnO	0.00	0.00	0.00
Total		100.00	100.00

high vacuum		13:58:42	
Oxide:	Net	Wt%	At%
Na2O	79.51	14.27	15.51
MgO	6.47	1.15	1.93
Al2O3	171.49	26.08	17.24
SiO2	317.12	57.21	64.17
CaO	7.39	0.77	0.93
Fe2O3	2.66	0.52	0.22
Total		100.00	100.00

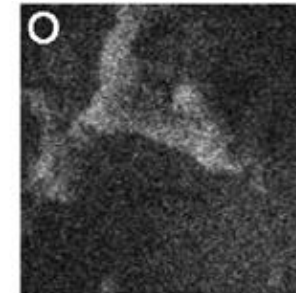
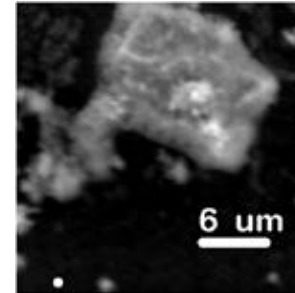
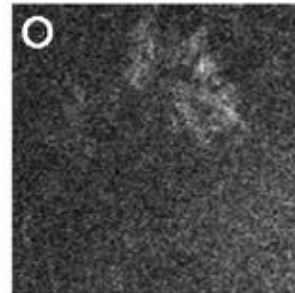
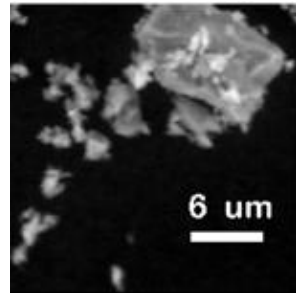
High Vac

Low Vac results (gas compensated) are close to those of the HV analysis.

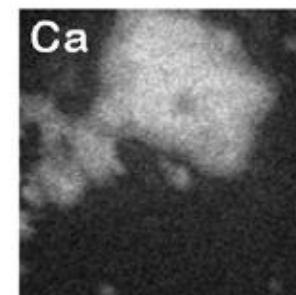
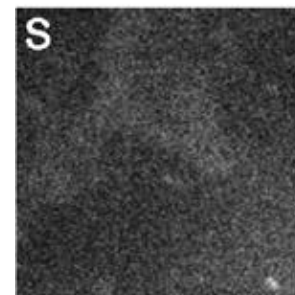
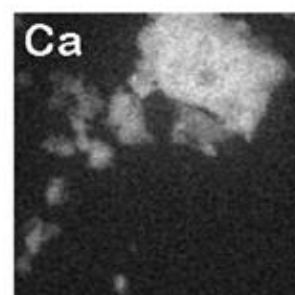
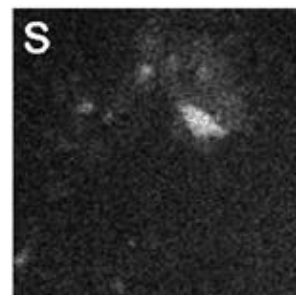
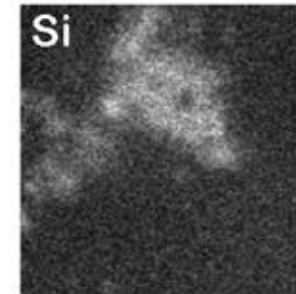
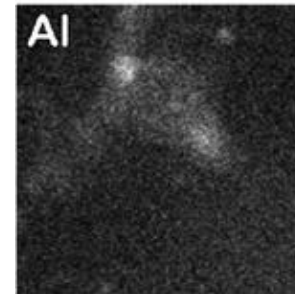
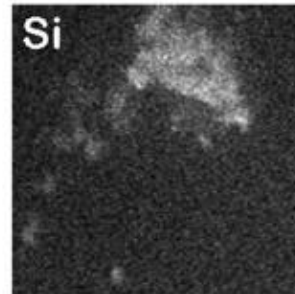
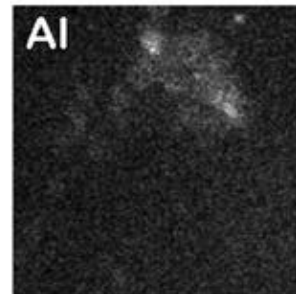


X-ray Mapping: Curing cement

Dry



Wet





Summary - Skirt Effect

➤ **Imaging:**

- **No problem; skirt adds a nearly uniform background**
- **Resolution is defined by central probe**

➤ **X-ray analysis:**

- **quantification is difficult but possible**
- **mapping stillz possible**

To reduce skirt effect:

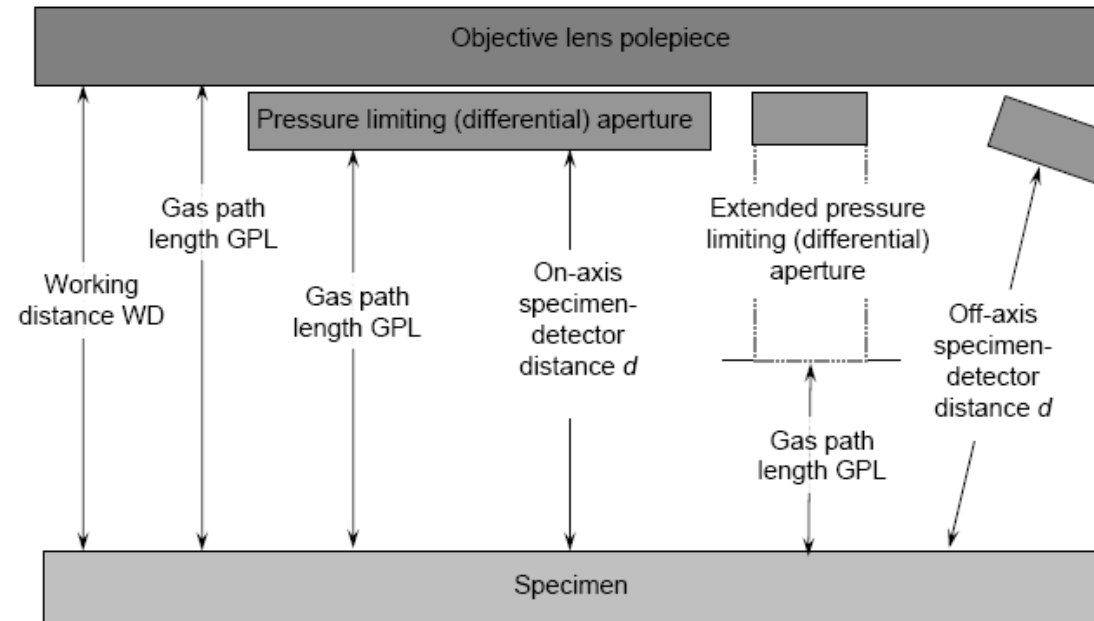
- **use short gas path (a special cone attached to pole piece)**
- **use high acceleration voltage**
- **use low pressure (0.3 - 0.4 torr)**

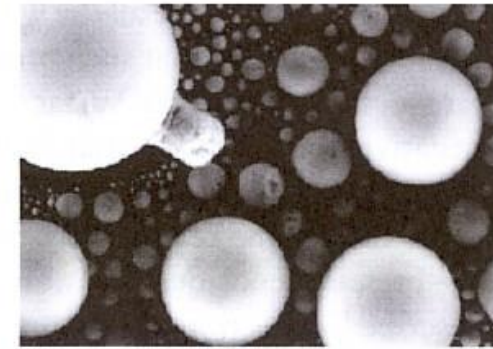
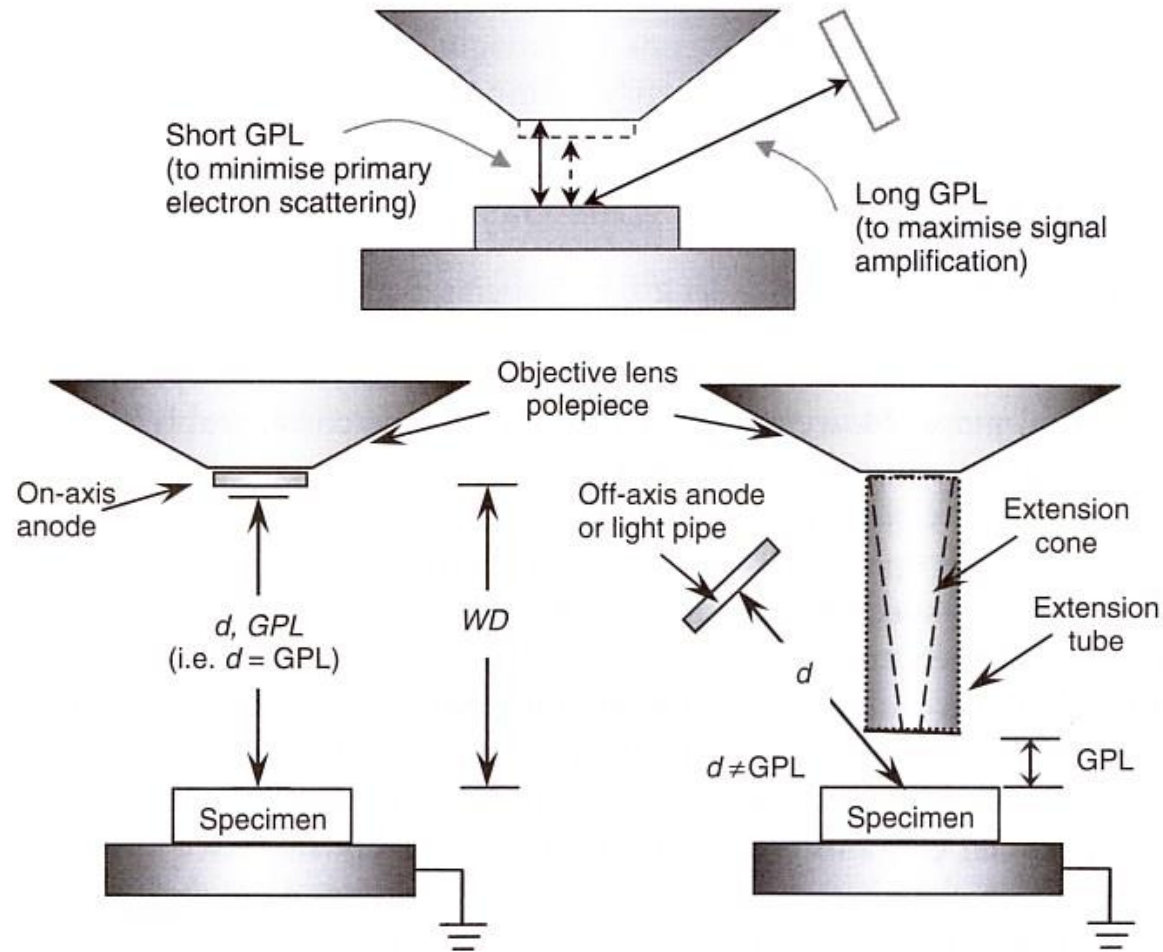
Trade-offs between variables:

- **Pressure**
- **Working distance**
- **Type of gas**
- **Energy of primary electrons**
- **Resolution**
- **Charge control**

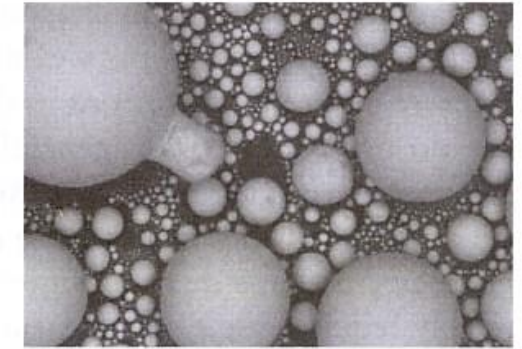


Primary electron scattering - definition of WD, d and GPL

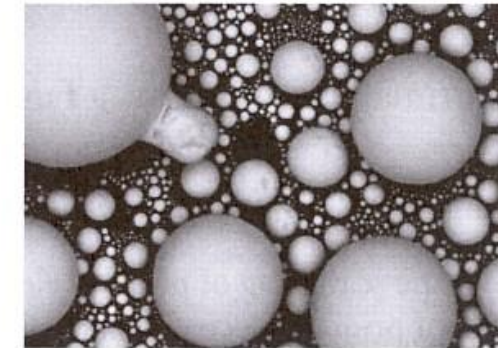




(a)

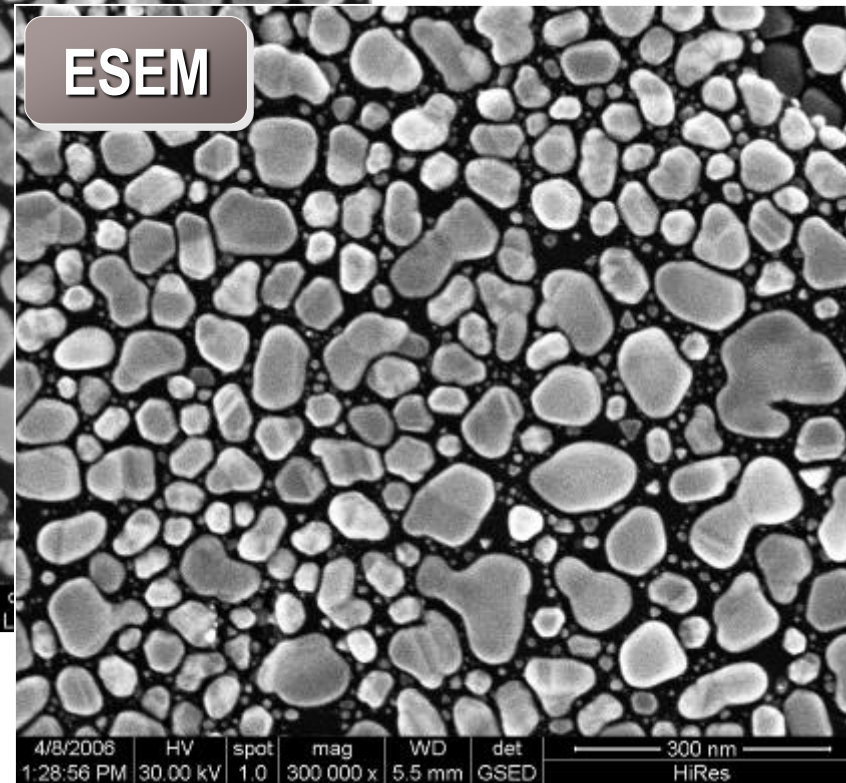
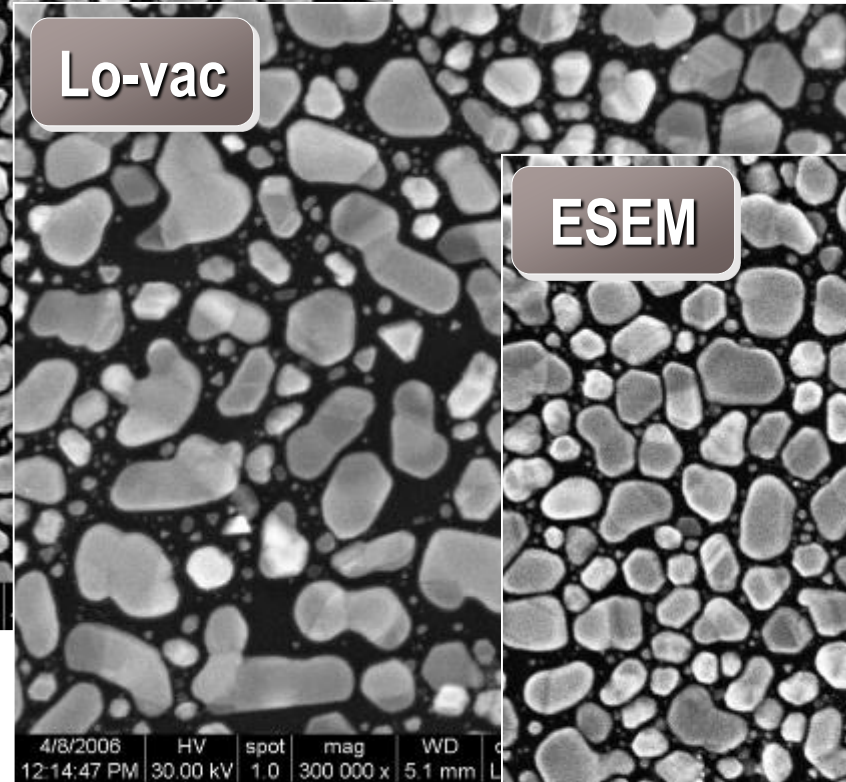
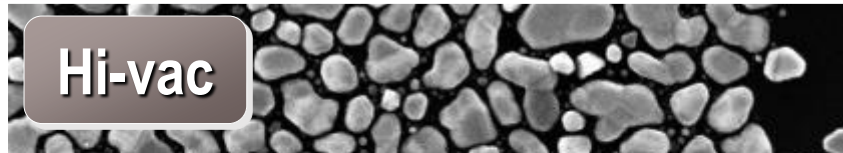


(b)



(c)

Figure 4.12 Backscattered electron images to show the effect of (a) short working distance, short gas path length (3 mm), (b) long working distance, long gas path length (10.5 mm) and (c) long working distance (10.5 mm), short gas path length (3 mm). Notice how in (c) the contrast and signal-to-noise have improved. Imaged in nitrogen gas with primary beam energy $E_0 = 20$ keV. Horizontal field width = $255 \mu\text{m}$. Images courtesy of Ken Robinson, Carl Zeiss SMT Ltd



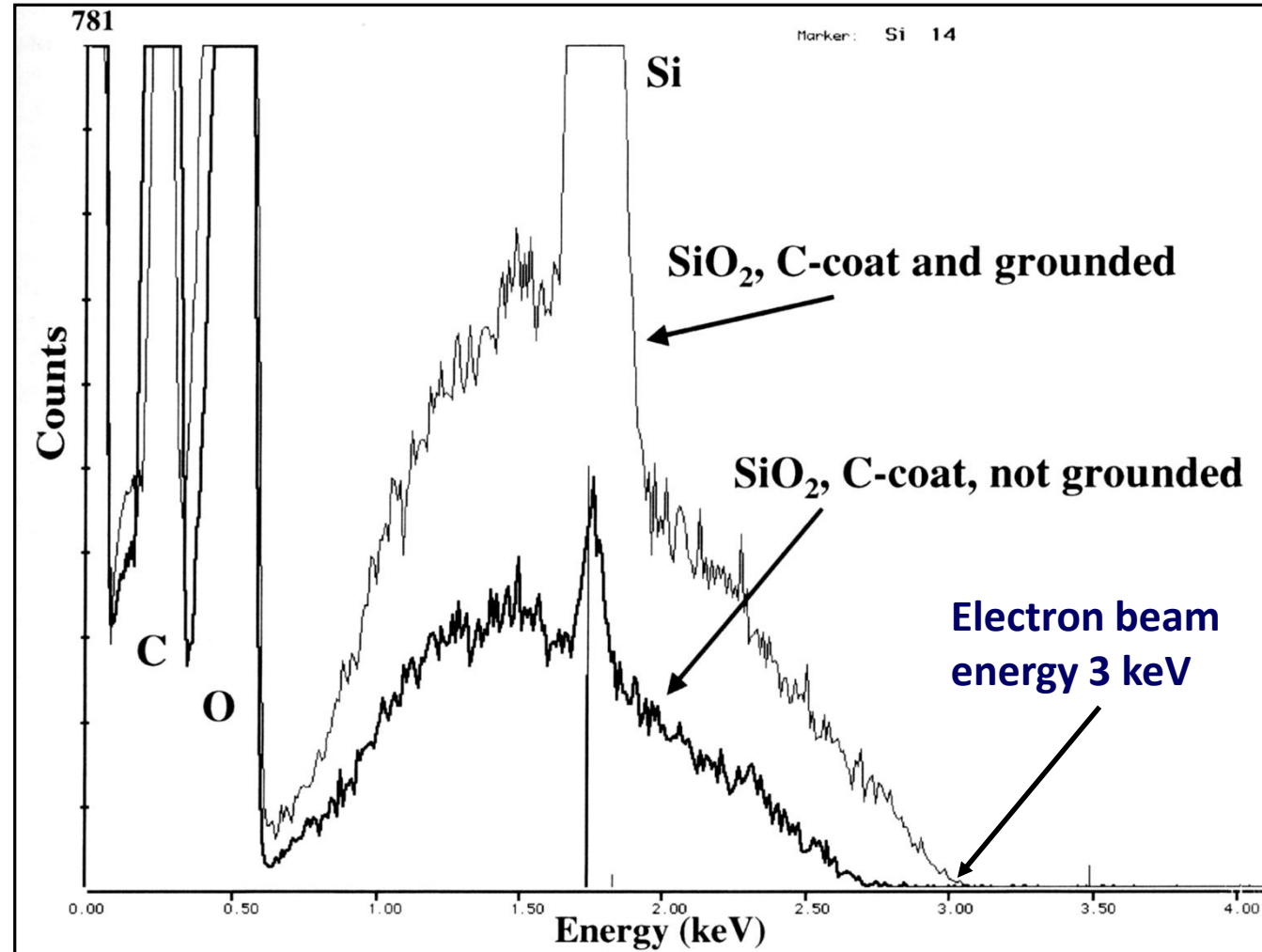
Imaging:

No problem; skirt adds a nearly uniform background

Resolution is defined by central probe



Charge control by Duane – Hunt limit





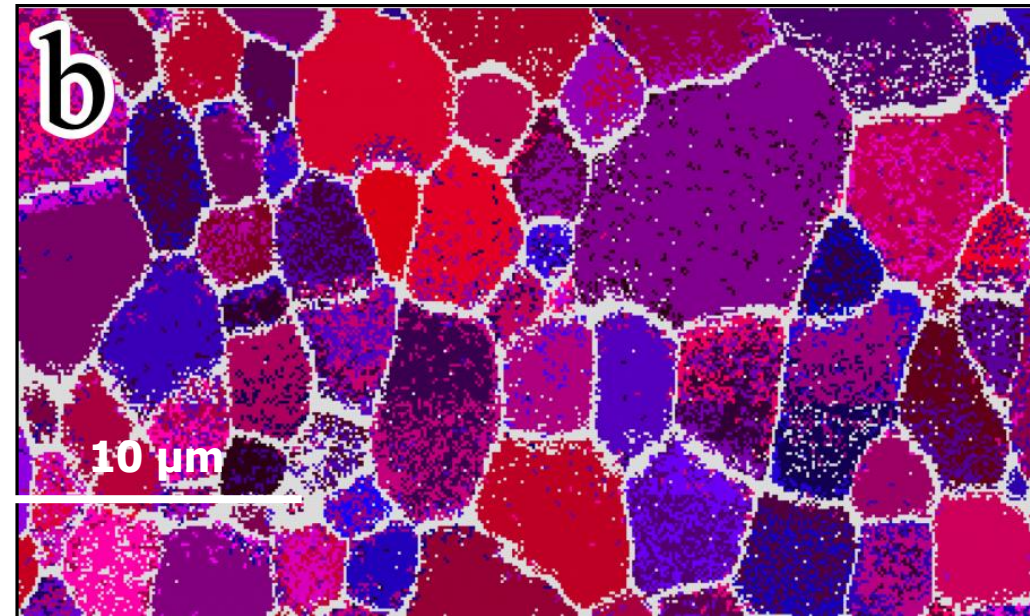
EBSD from non-conductive samples



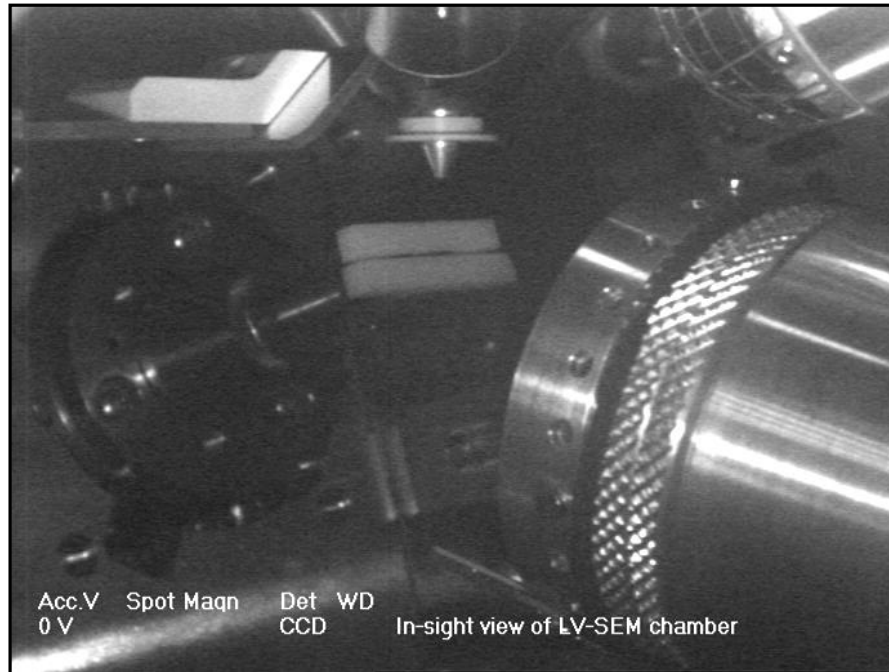
Orientation maps acquired from non-conductive sample (cubic ZrO_2)



a) C-SEM – electric charge
non compensated
All Euler map



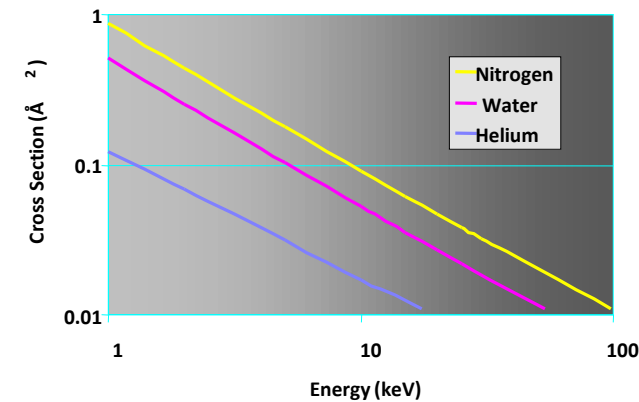
b) VP-SEM – electric charge
compensated
All Euler map



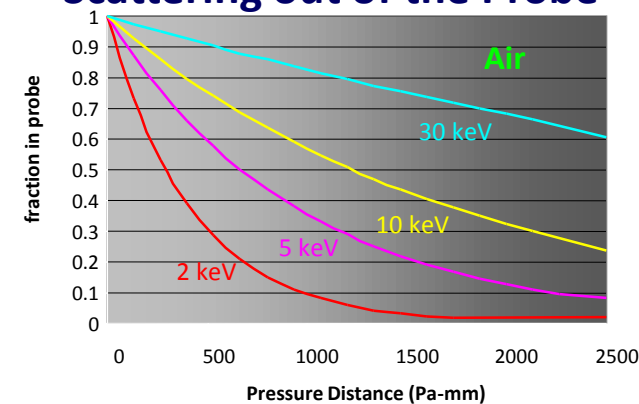
Trade-offs between several variables:

- Pressure
- Working distance
- Type of gas
- Energy of primary electrons
- Resolution
- Charge control

Elastic Cross Section



Scattering out of the Probe



Rule of thumb:

Use as low pressure and small polepiece-specimen distance as possible!



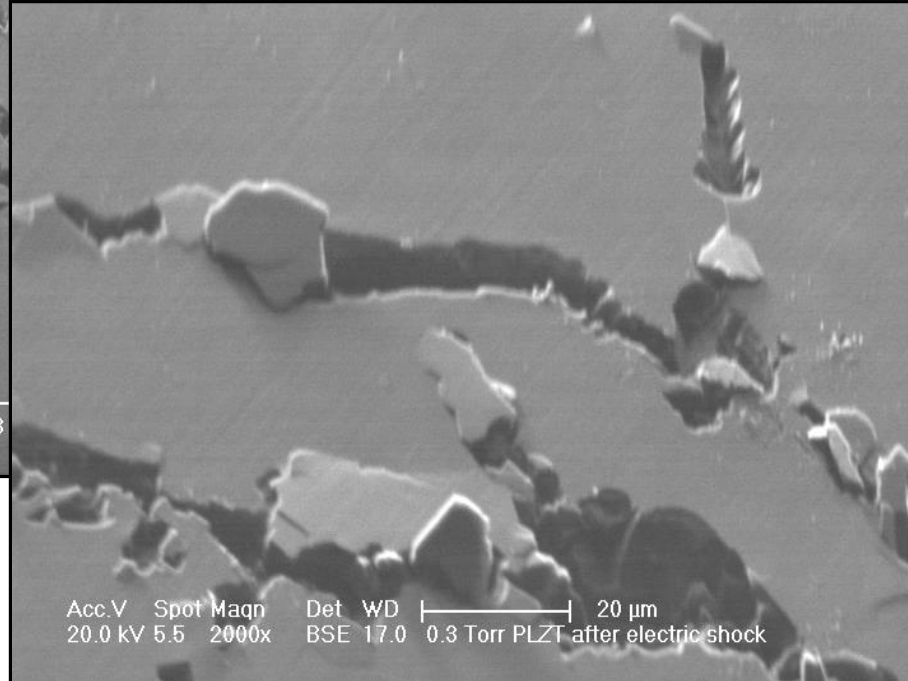
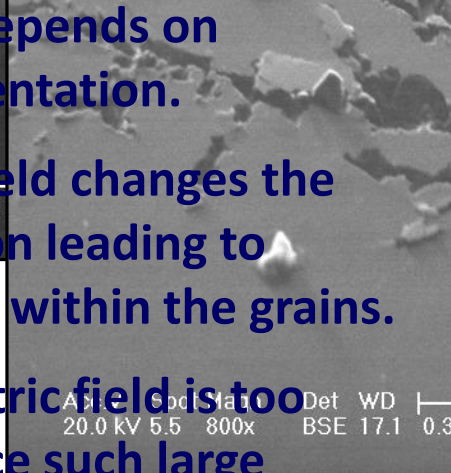
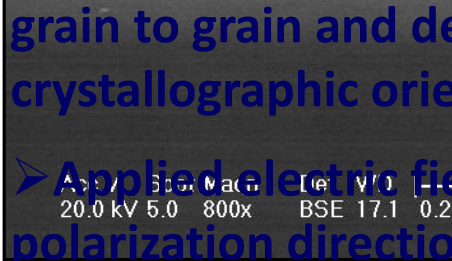
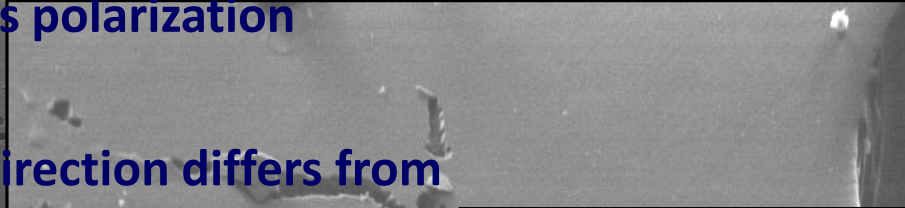
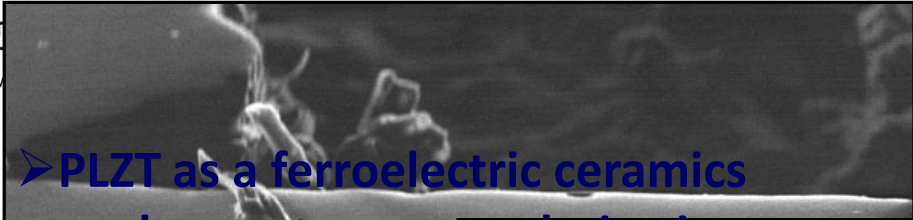
What happens when the charge is not compensated

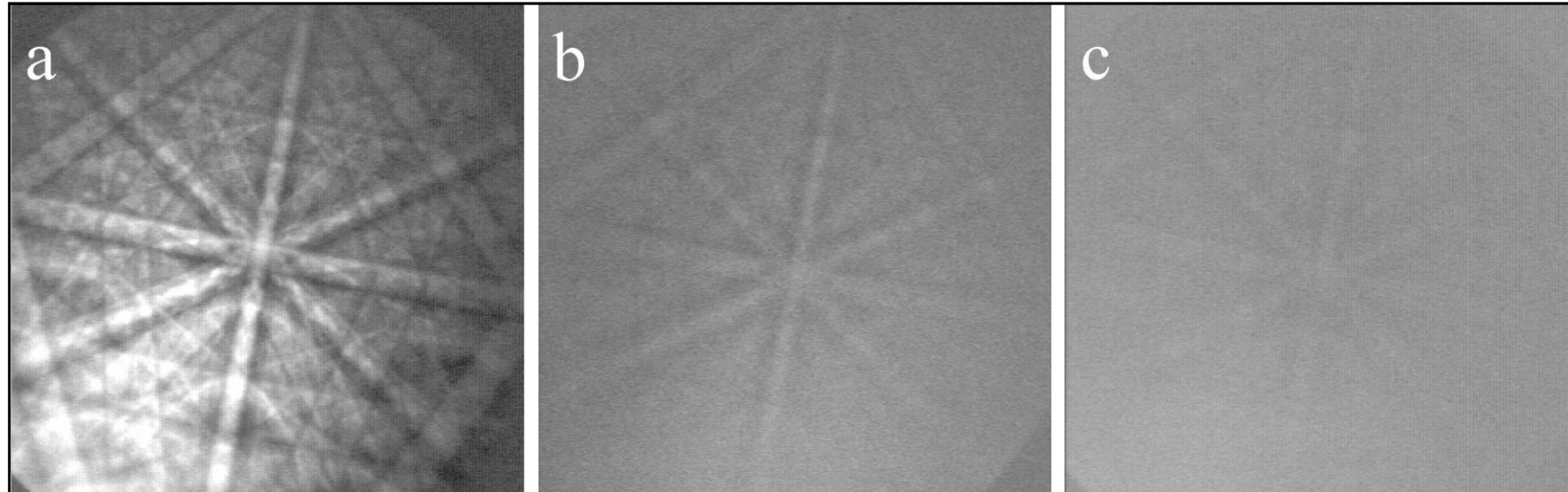
➤ PLZT as a ferroelectric ceramics reveals spontaneous polarization within the grains.

➤ The polarization direction differs from grain to grain and depends on crystallographic orientation.

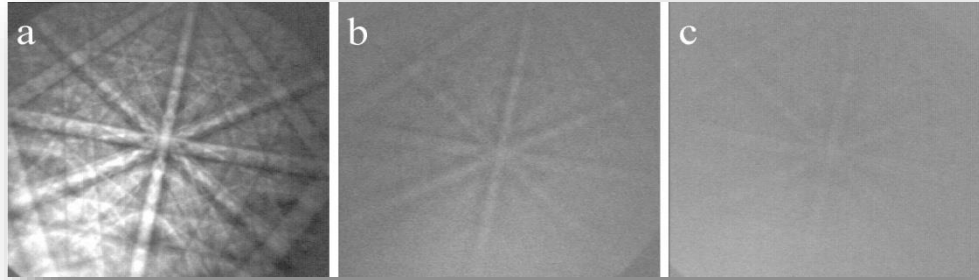
➤ Applied electric field changes the polarization direction leading to mechanical stresses within the grains.

➤ If the applied electric field is too strong, it can produce such large mechanical stress and corresponding strain within the grains that they can be pulled out and cracks are formed.

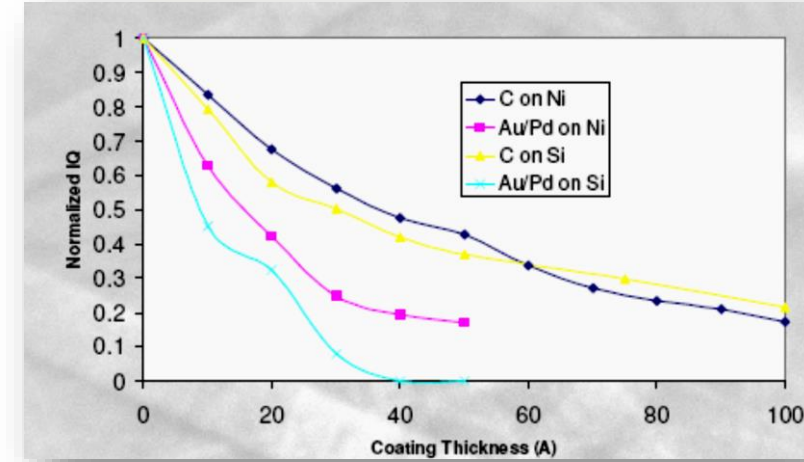




Effect of coating on diffraction pattern from NiO monocrystal
a) no coating, b) carbon coating, c) gold coating

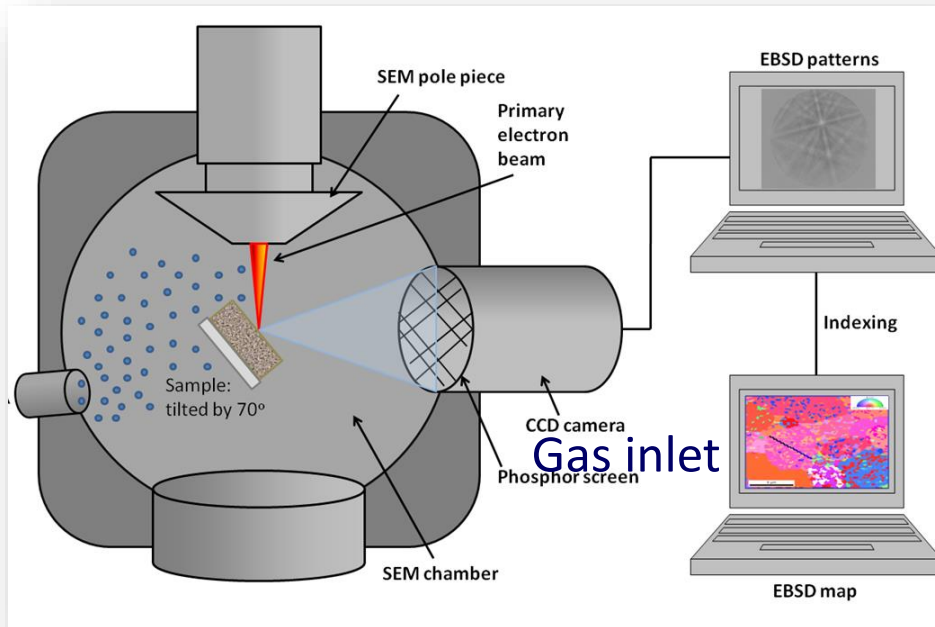


Effect of coating on diffraction pater quality
a) no coating – VP-SEM, b) carbon coating, c) gold coating



Courtesy of EDAX

Low backscatter yield of the ceramics due to low average atomic number !!!

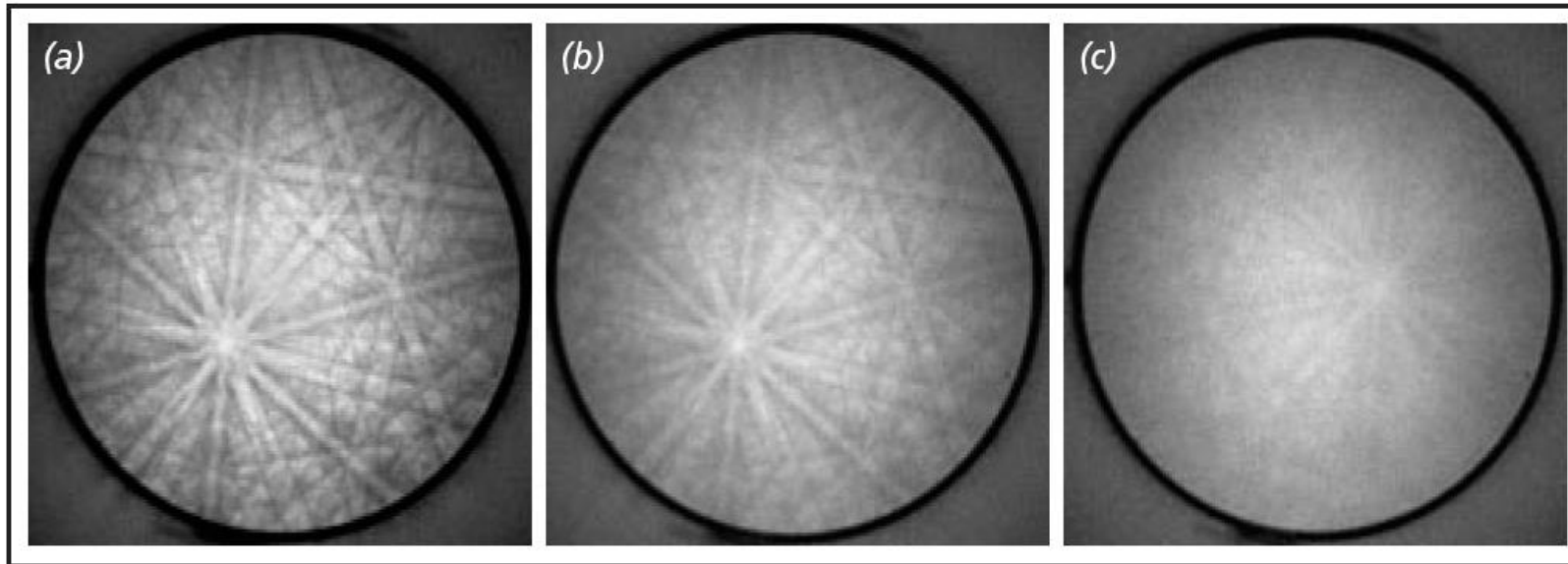


Variable
Pressure SEM
seems to be a
good solution
to overcome
charging
problems!

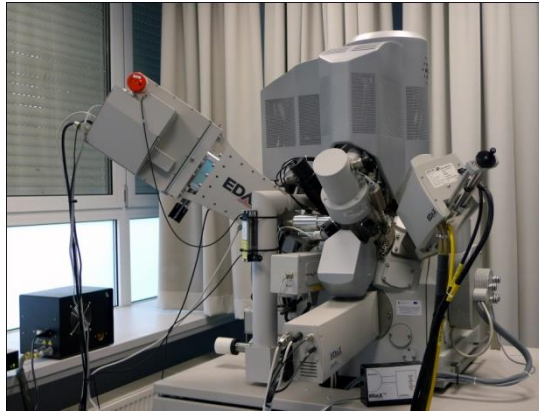
Project WND-POWR.03.02.00-00-I043/16

International interdisciplinary PhD Studies in Materials Science with English as the language of instruction

Project co-financed by the European Union within the European Social Funds

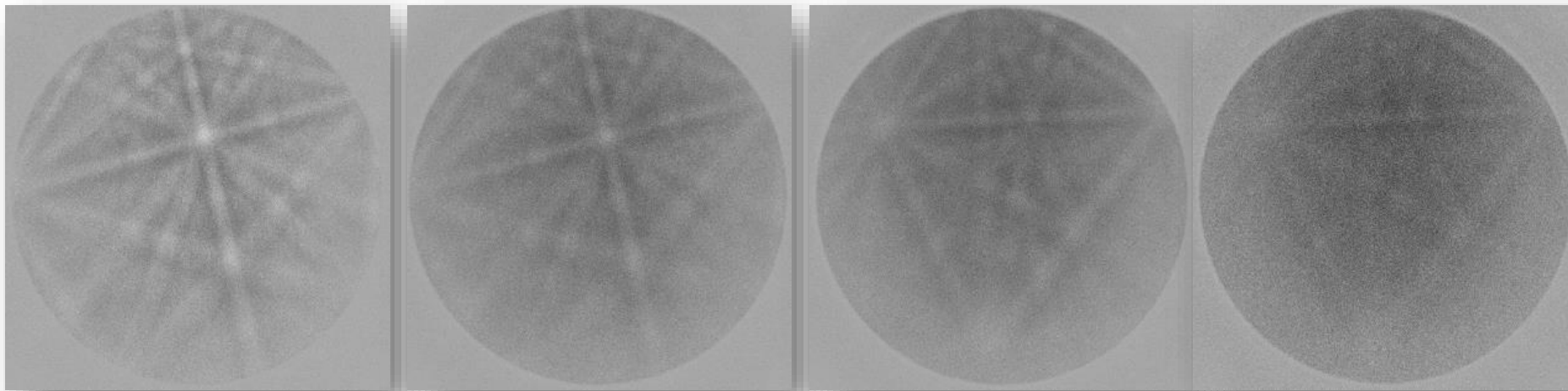
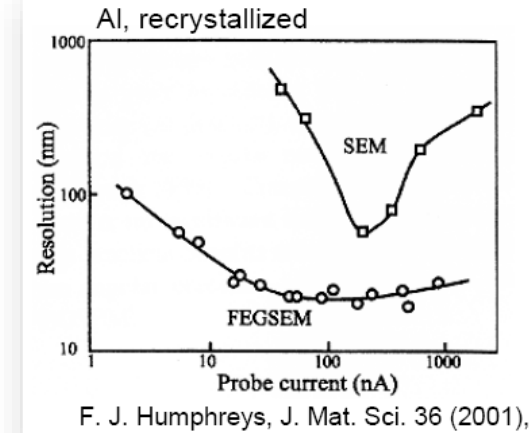


Effect of vacuum in SEM chamber on diffraction pattern from Pt
a) 0.05 Torr, b) 0.5 Torr, c) 1.0 Torr



FEGSEM

- a huge electrical charging
due high beam current
- mechanical and electron
beam drift

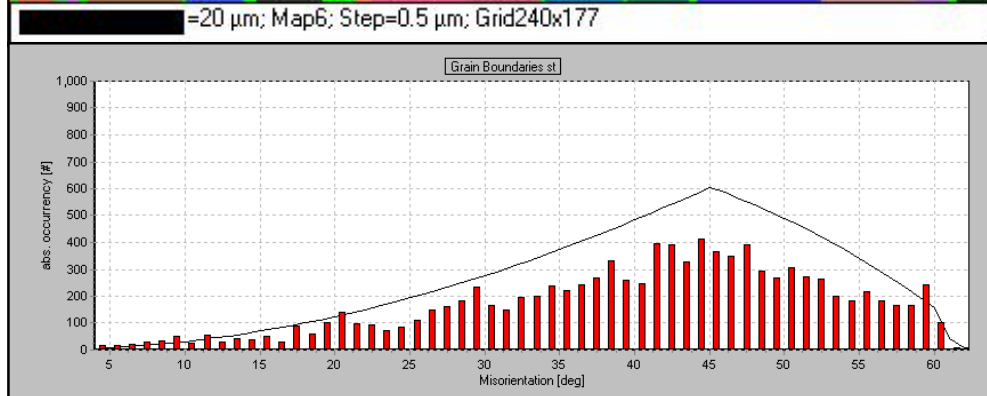
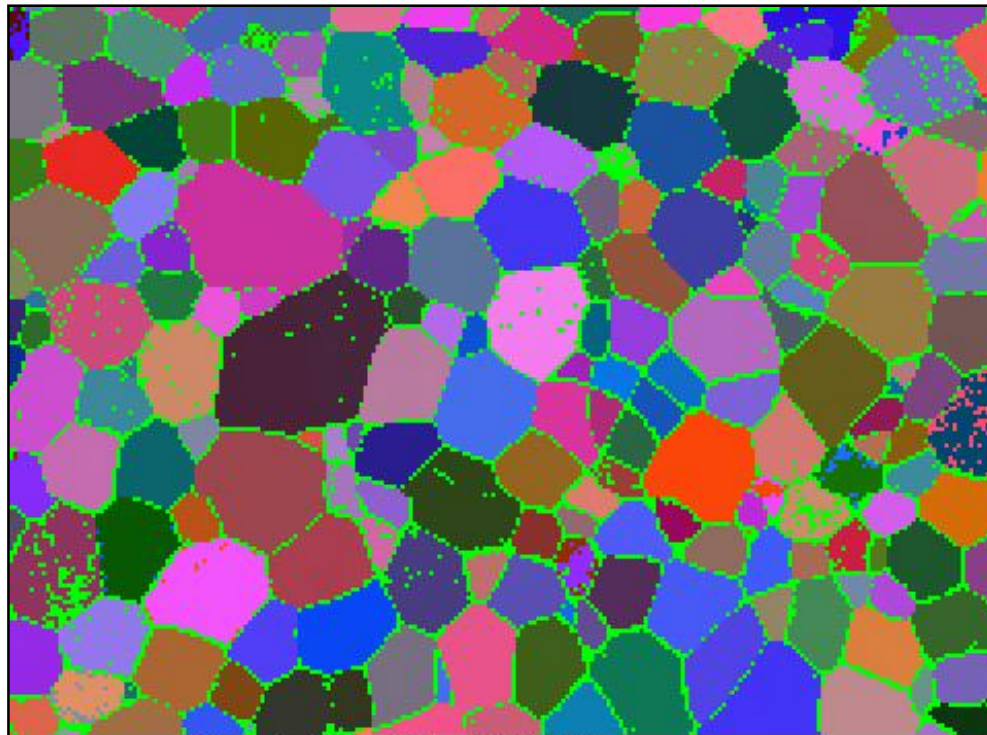


EBSD pattern recorded from the PLZT ceramics in FEGSEM at different pressures:
a) 50 Pa, b) 70 Pa, c) 90, d) 130 Pa



In order to achieve the best electron backscatter diffraction quality in low vacuum conditions, there are some parameters in the SEM which need to be adjusted, mainly:

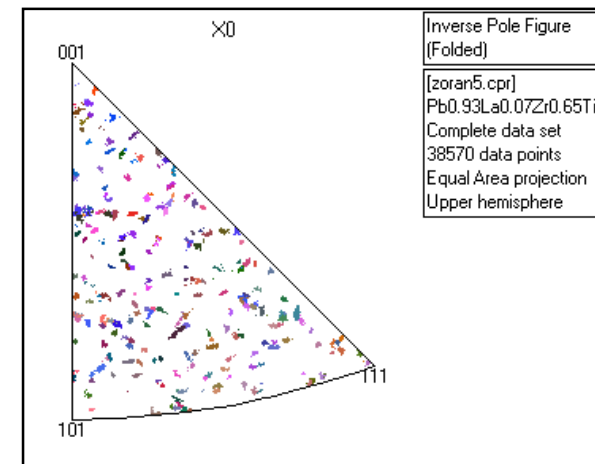
- beam energy and current,
- gas path length,
- gas pressure,
- dwell time.



PLZT ceramics

$Pb_{1-3x/2}La_xZr_{0.65}Ti_{0.35}O_3$ for $x = 0.08$
(denoted as PLZT 8/65/35)

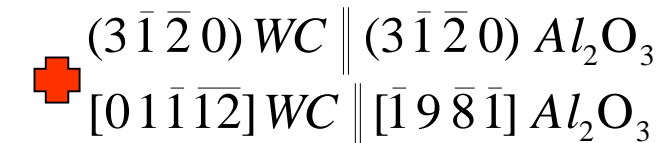
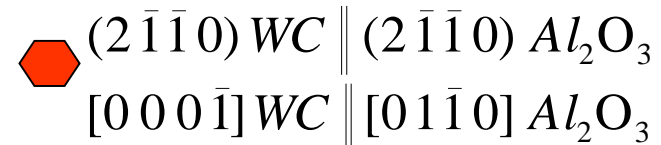
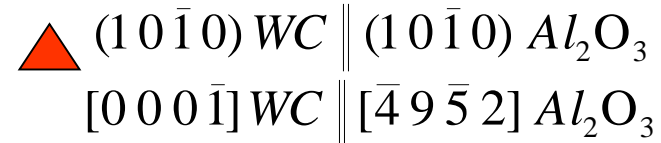
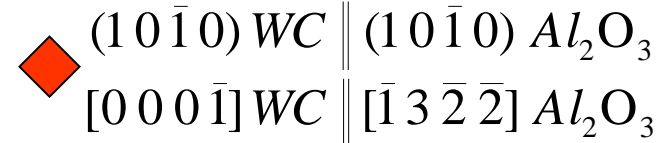
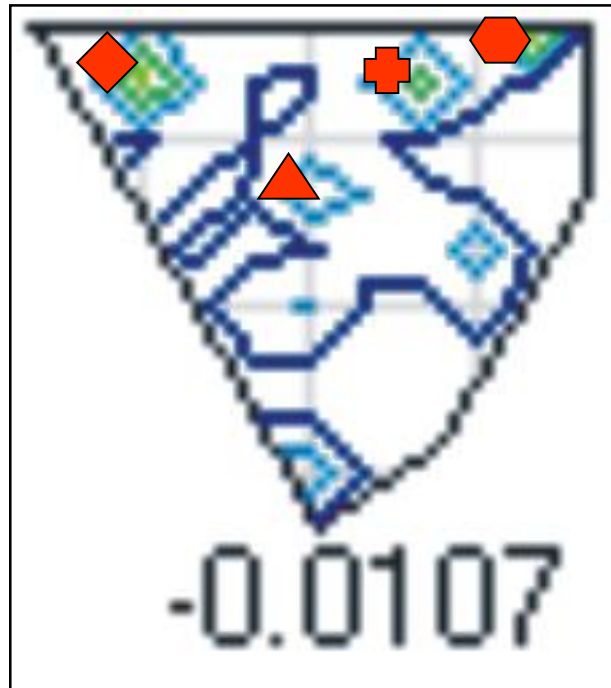
**91% of diffraction patterns
solved!!! All Euler map**



**Inverse pole figure shows
randomly oriented grains**

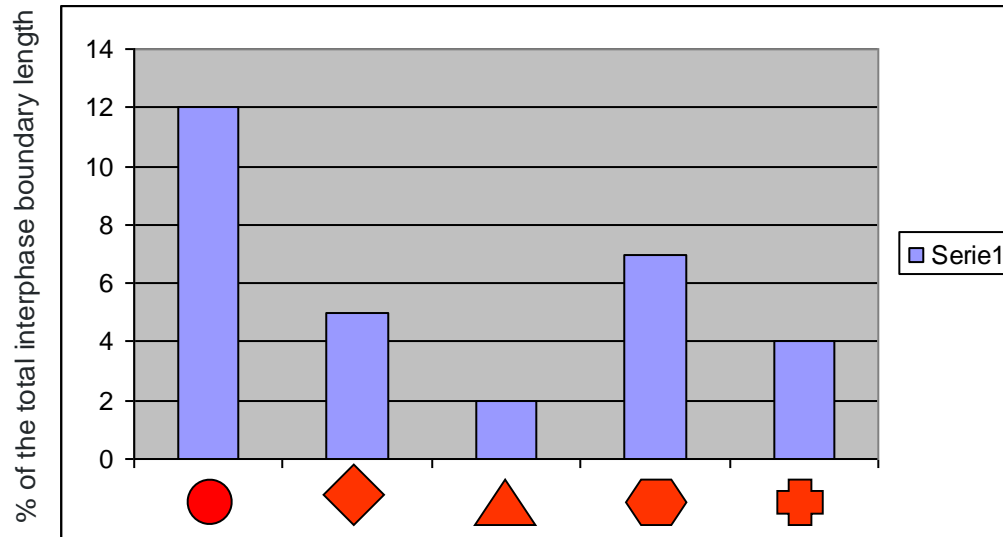


Misorientation Distribution Function (Al_2O_3/WC)



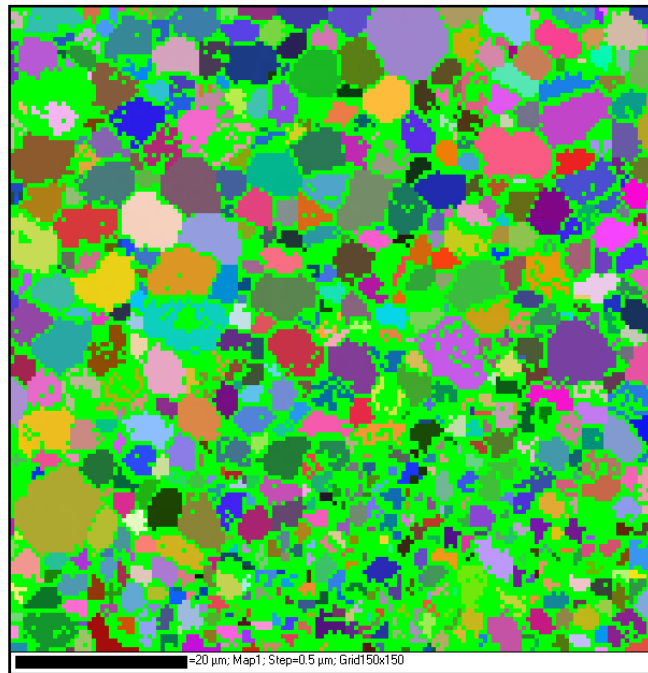


Misorientation Distribution Function (Al_2O_3/WC)

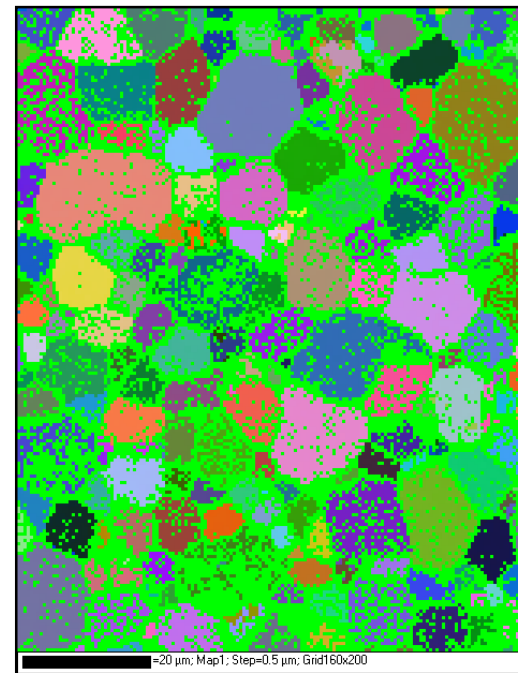


The crystallographic relationships correspond respectively to: 12%, 5%, 2%, 7% and 4% of the total Al_2O_3/WC interphase boundary length.

- $(0001) WC \parallel (0001) Al_2O_3$
 $[11\bar{2}0] WC \parallel [10\bar{1}0] Al_2O_3$
- ◆ $(10\bar{1}0) WC \parallel (10\bar{1}0) Al_2O_3$
 $[000\bar{1}] WC \parallel [\bar{1}3\bar{2}\bar{2}] Al_2O_3$
- ▲ $(10\bar{1}0) WC \parallel (10\bar{1}0) Al_2O_3$
 $[000\bar{1}] WC \parallel [\bar{4}9\bar{5}2] Al_2O_3$
- ⬡ $(2\bar{1}\bar{1}0) WC \parallel (2\bar{1}\bar{1}0) Al_2O_3$
 $[000\bar{1}] WC \parallel [01\bar{1}0] Al_2O_3$
- ⊕ $(3\bar{1}\bar{2}0) WC \parallel (3\bar{1}\bar{2}0) Al_2O_3$
 $[01\bar{1}\bar{1}\bar{2}] WC \parallel [\bar{1}9\bar{8}\bar{1}] Al_2O_3$



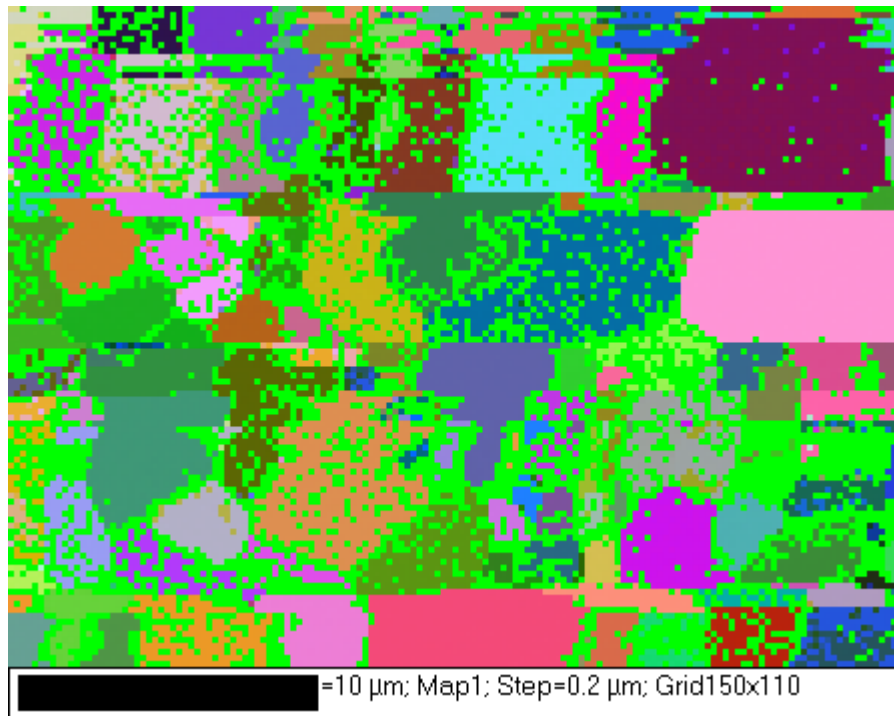
Al₂O₃ (H₂O pressure - 0.4 mbar)



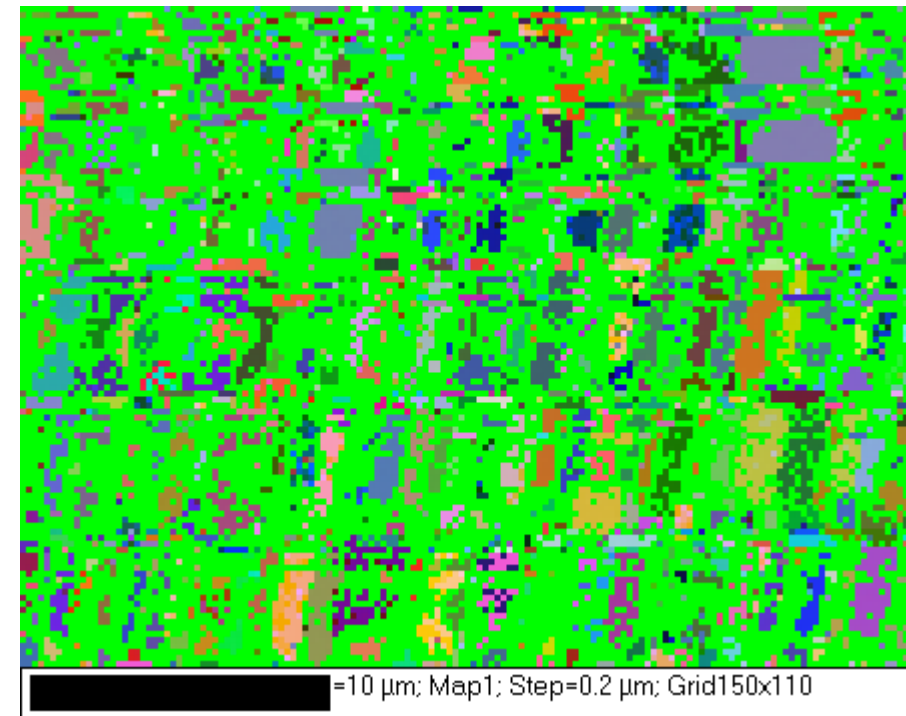
Al₂O₃ (pressure H₂O - 1.33 mbar)



When the pressure is too low...



Coarse grained Al_2O_3



Fine grained Al_2O_3



Thank you for attention

Project WND-POWR.03.02.00-00-1043/16

International interdisciplinary PhD Studies in Materials Science with English as the language of instruction

Project co-financed by the European Union within the European Social Funds